

Molecular Dynamics Simulations on the Austenite-Martensite Phase Transition in Fe_7Pd_3

CompPhys17, November 30 – December 2, 2017

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Motivational Introduction

Shape Changing Effects

- Thermal Shape Memory Effect
 - Invar Effect
- Ferromagnetic Shape Memory Effect

Mechanical Properties

- High and low temperature solid state phases
- Martensitic transformations
- Highly ductile
- Magnetically anisotropic
- High Currie temperature (720 K)

Biocompatibility

- No foreign body reactions
- Good adhesion properties



Functional Surfaces

Applications

- Mechanically active coatings for prosthesis
 - Vascular Grafts
- Magnetically controllable actuators for medical applications [*]

[*] K. Ullakko, L. Wendell, A. Smith, P. Müllner, and G. Hampikian, Smart Mater. and Struct. 21, 115020 (2012).

Outline

1. Martensitic Transformations
2. Potential for Molecular Dynamics Simulations
3. Temperature Dependent Ground-States
4. Strain Induced Transformation
5. Conclusion

Martensitic Transformations

- Transformations between high temperature, high symmetric austenite and low temperature, lower symmetric martensite phase

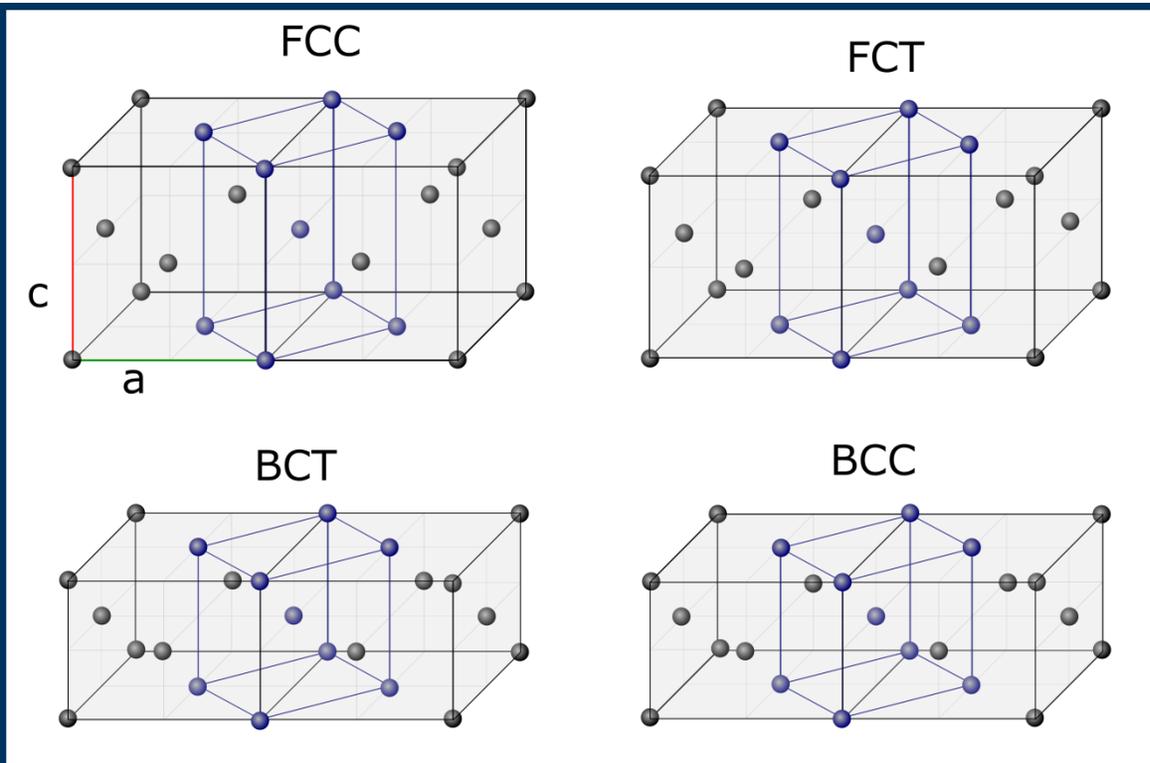
Martensitic Transformations

- Transformations between high temperature, high symmetric austenite and low temperature, lower symmetric martensite phase
- Changes of the lattice structure and shape

Martensitic Transformations

- Transformations between high temperature, high symmetric austenite and low temperature, lower symmetric martensite phase
- Changes of the lattice structure and shape
- Different transformation mechanisms proposed
 - Small ranges of atomic movement
 - Conservation of nearest neighbour relations

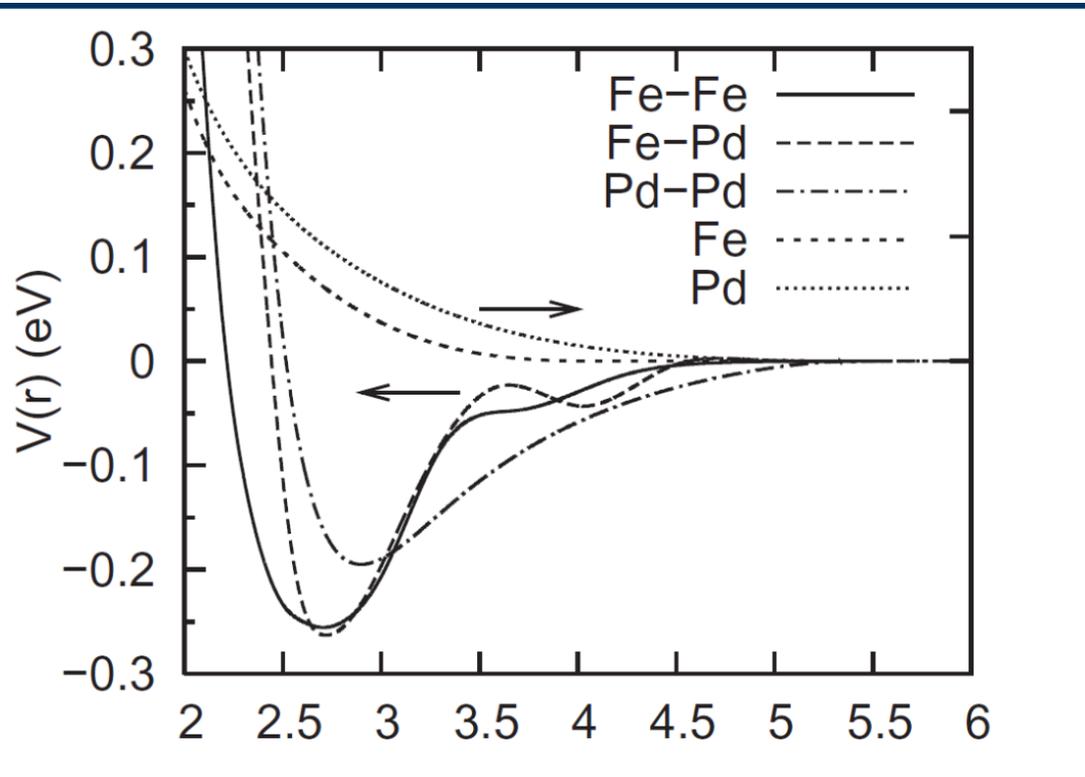
Martensitic Transformations



Bain Path

- Uniaxial Deformation parallel to the c-direction of the unit cell
- four stable structures corresponding to axis ratios:

Crystal Structure	Axis Ratio (c/a)
FCC	1
FCT	0.940
BCT	0.717
BCC	0.707



Potential for Molecular Dynamics Simulations

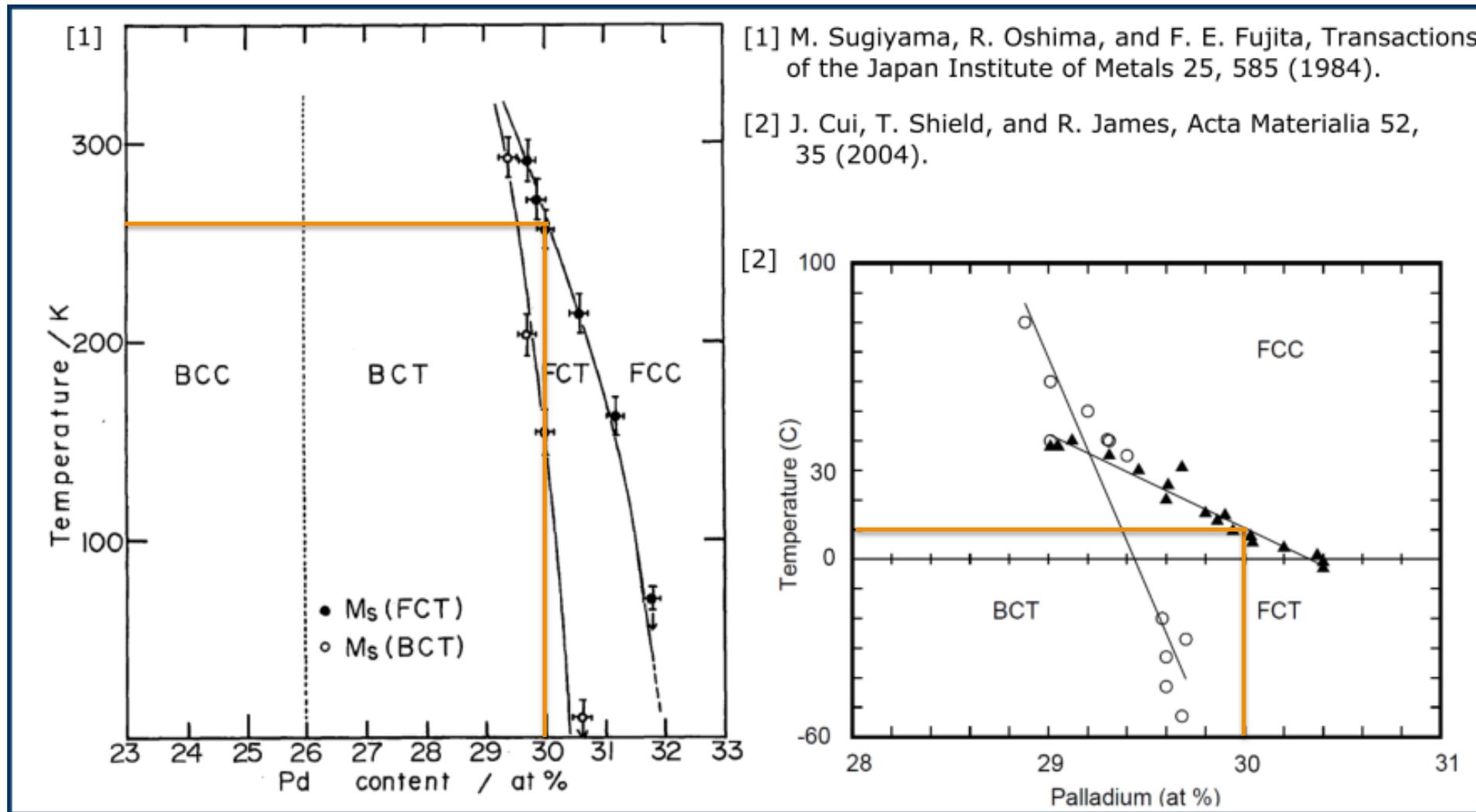
- Embedded Atom Method

$$E_{tot} = \sum_i F_i(\rho_i(R_i)) + \frac{1}{2} \sum_{i,j;j \neq i} \phi(R_{ij})$$

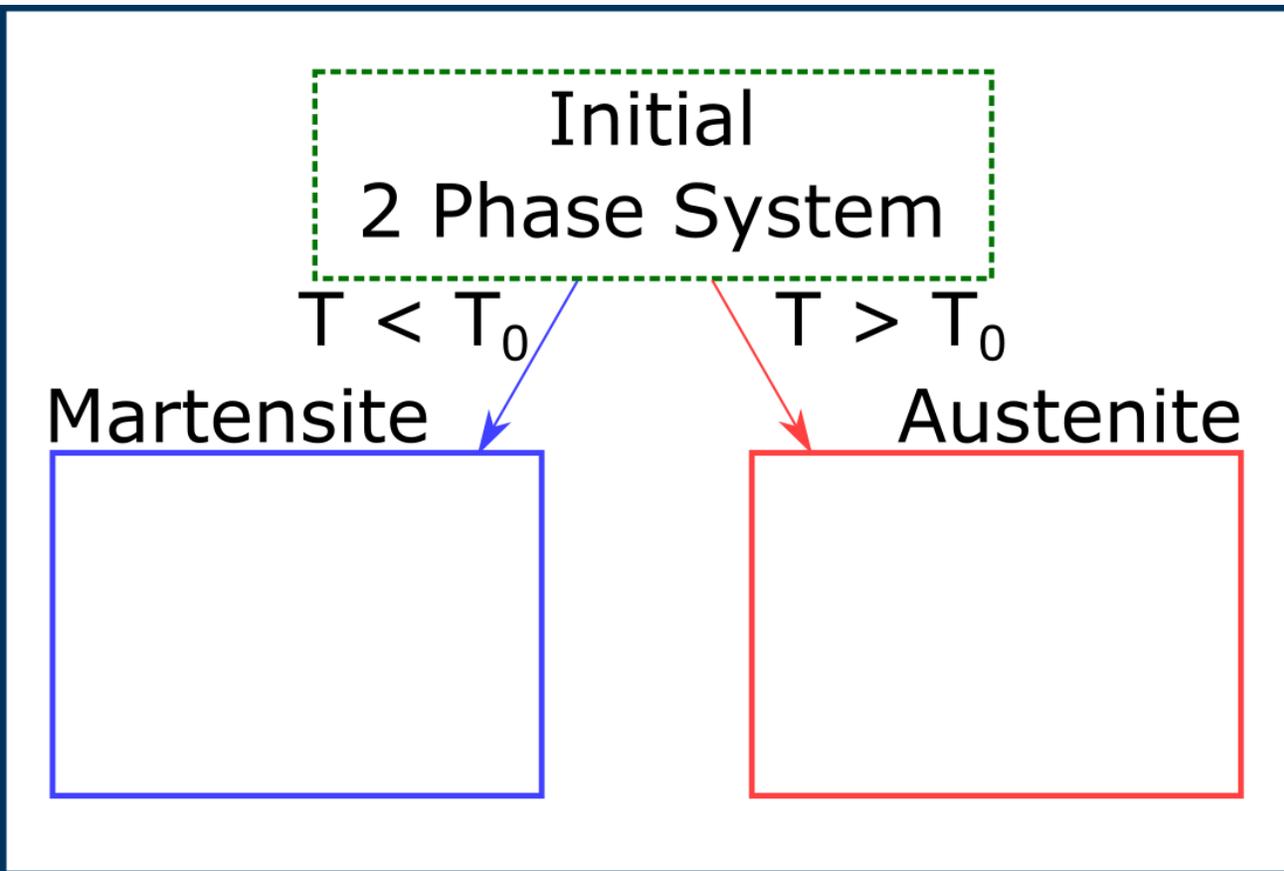
- Fit of the potential in order to adept Bain path properties observed in DFT simulations

[S G Mayr and A Arabi-Hashemi
2012 *New J. Phys.* **14** 103006]

Thermally Induced Phase Transition



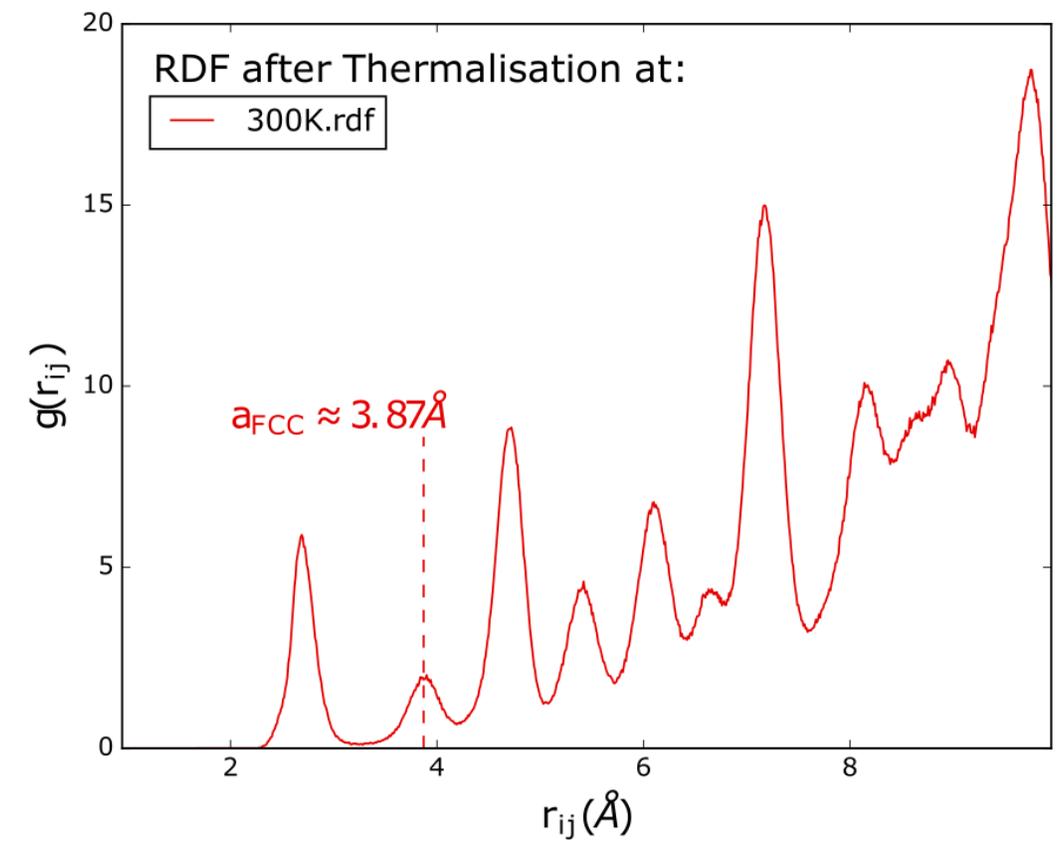
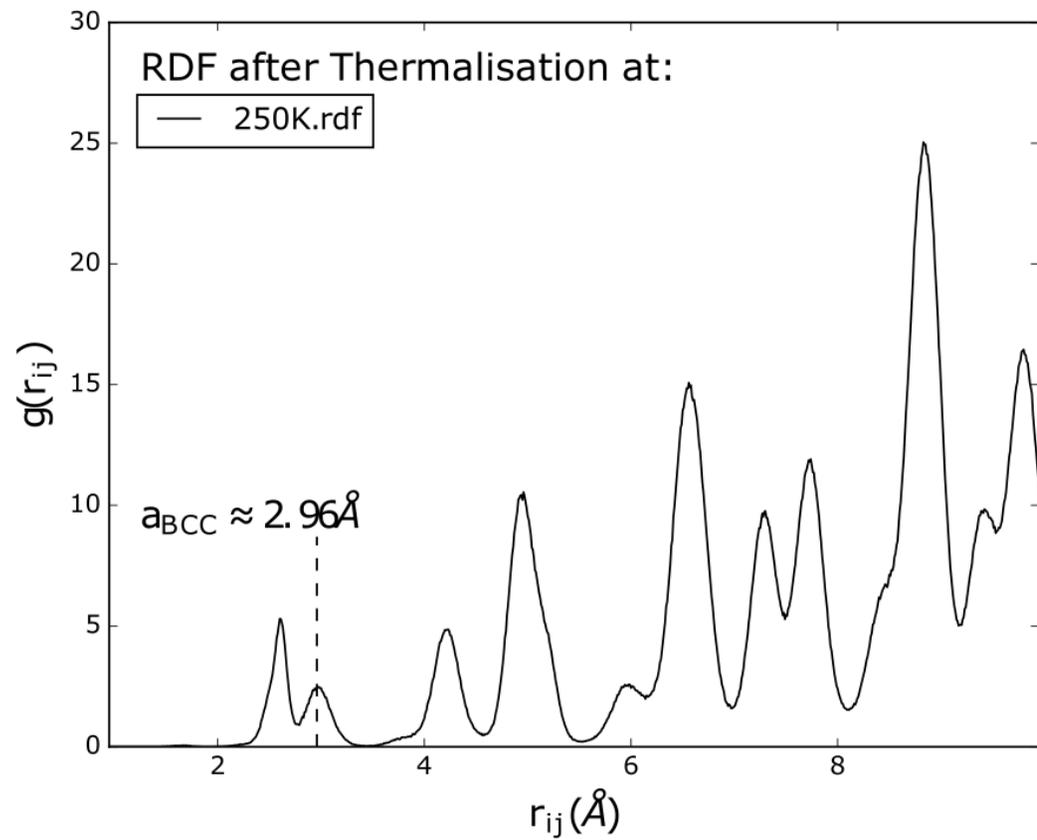
Thermally Induced Phase Transition



Resulting Phases

- Preparing a two phase system
- Depending on Temperature: two different equilibrium phases
- Compute the Radial Distribution Function (RDF)

Thermally Induced Phase Transition



Thermally Induced Phase Transition

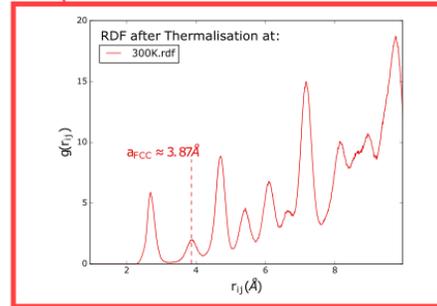
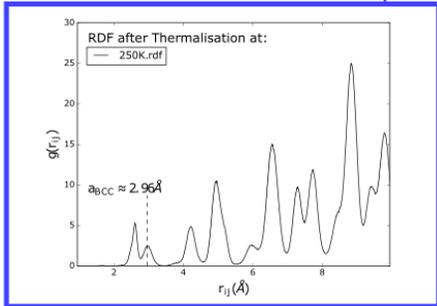
Initial
2 Phase System

$T < T_0$

$T > T_0$

Martensite

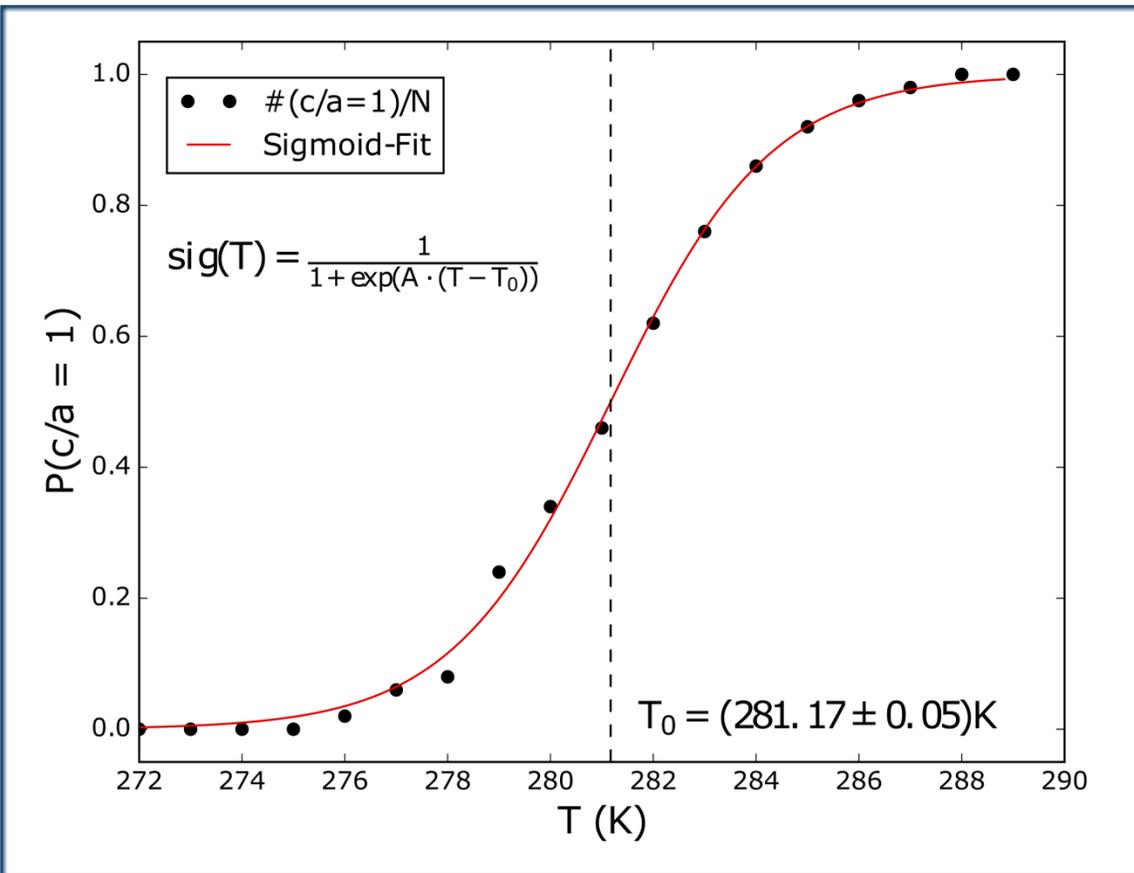
Austenite



Resulting Phases

- Providing 50 Temperature Sets for each Temperature to the two phase system
- Measure the relative frequentness of the *FCC* (austenite) phase compared to the *BCC* (martensite) phase

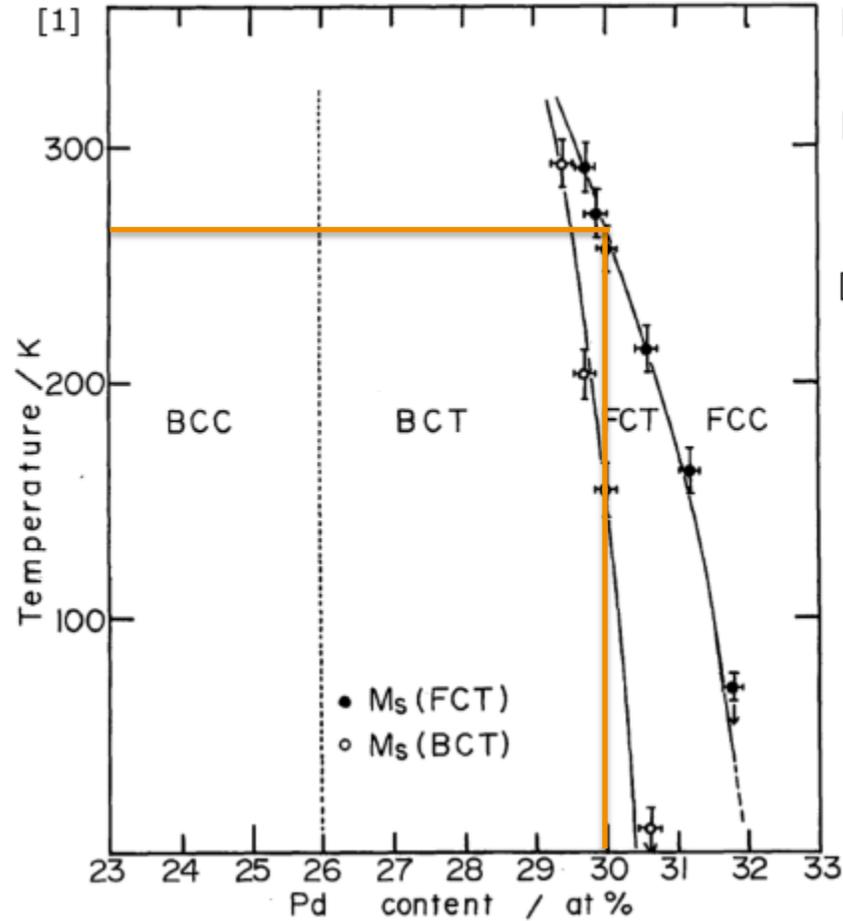
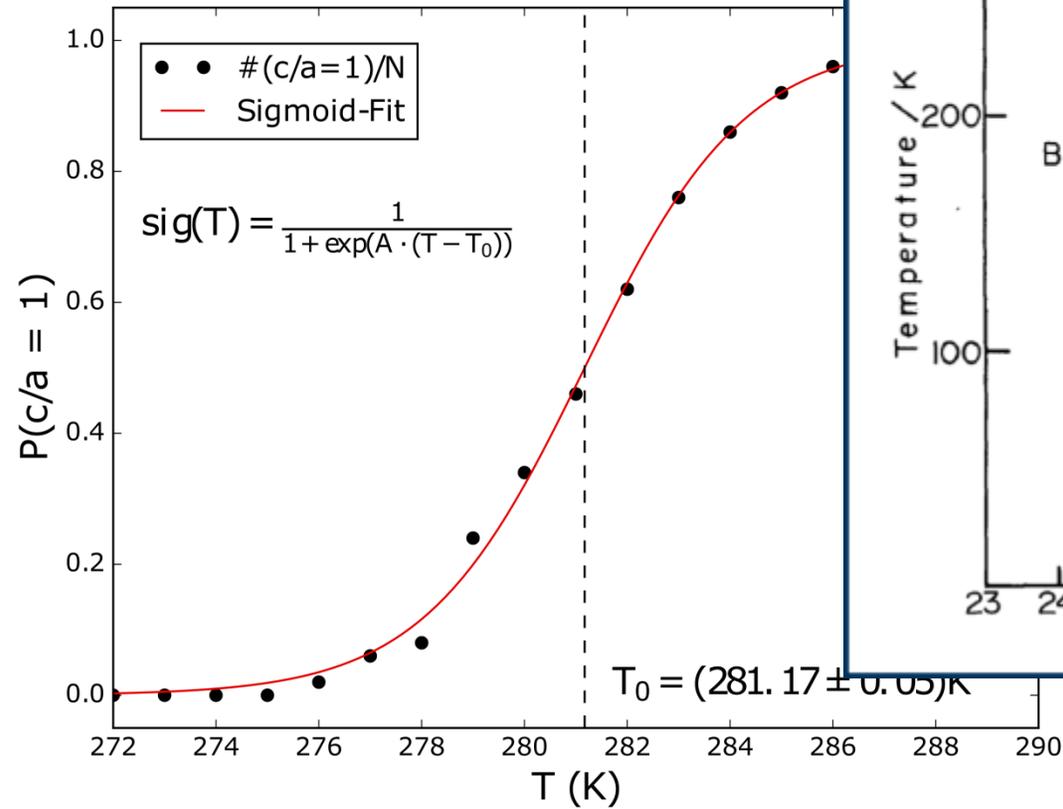
Thermally Induced Phase Transition



Resulting Phases

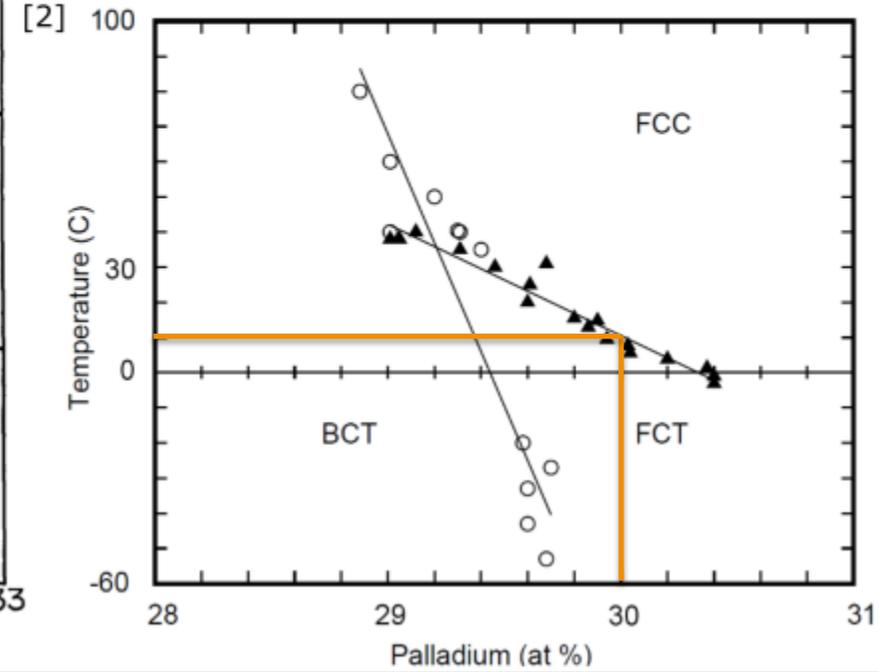
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Ther



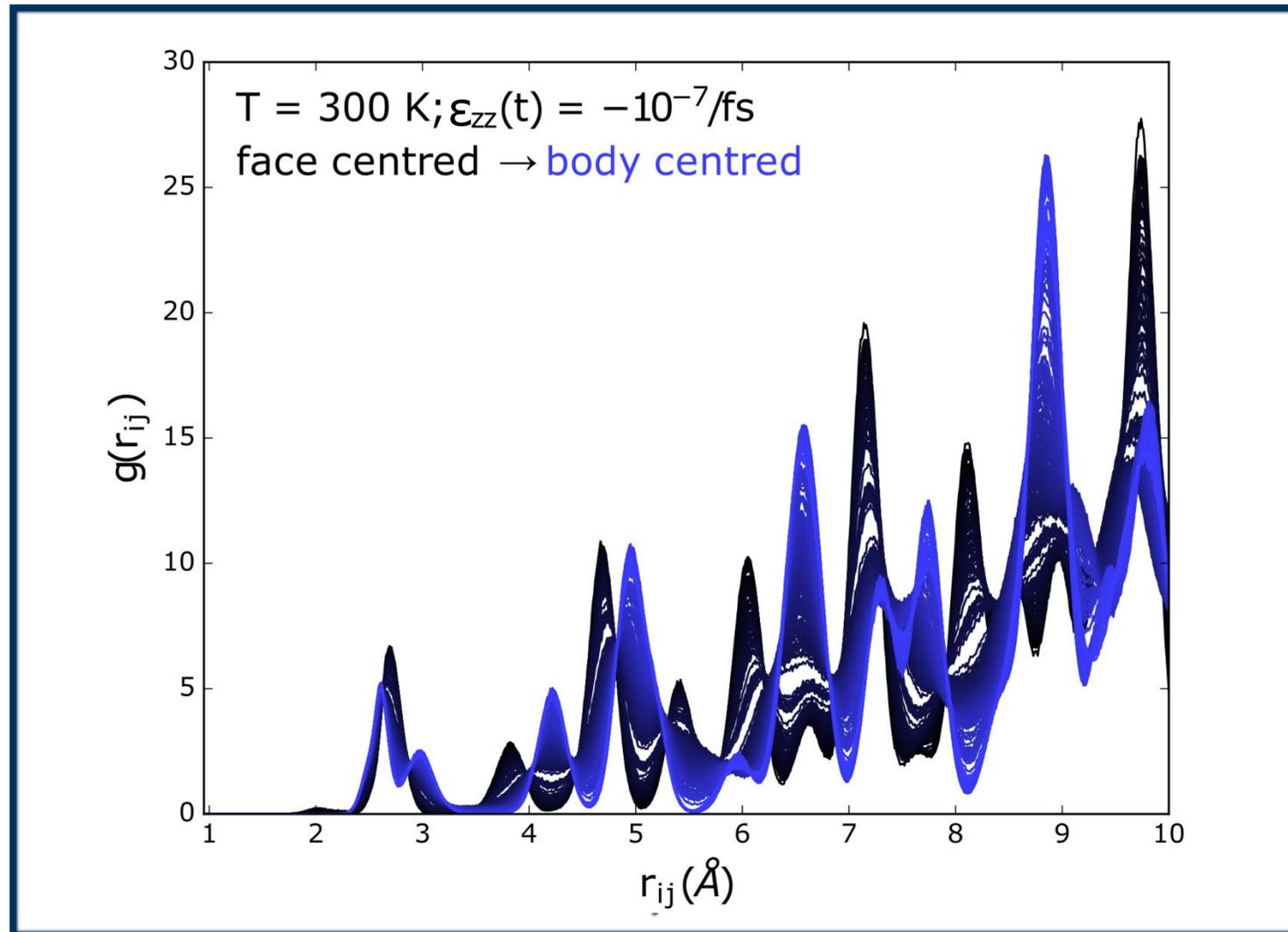
[1] M. Sugiyama, R. Oshima, and F. E. Fujita, Transactions of the Japan Institute of Metals 25, 585 (1984).

[2] J. Cui, T. Shield, and R. James, Acta Materialia 52, 35 (2004).



- Good accordance of the determined phase transition temperature $((281.17 \pm 0.05) \text{K})$ to literature values

Strain Induced Phase Transition



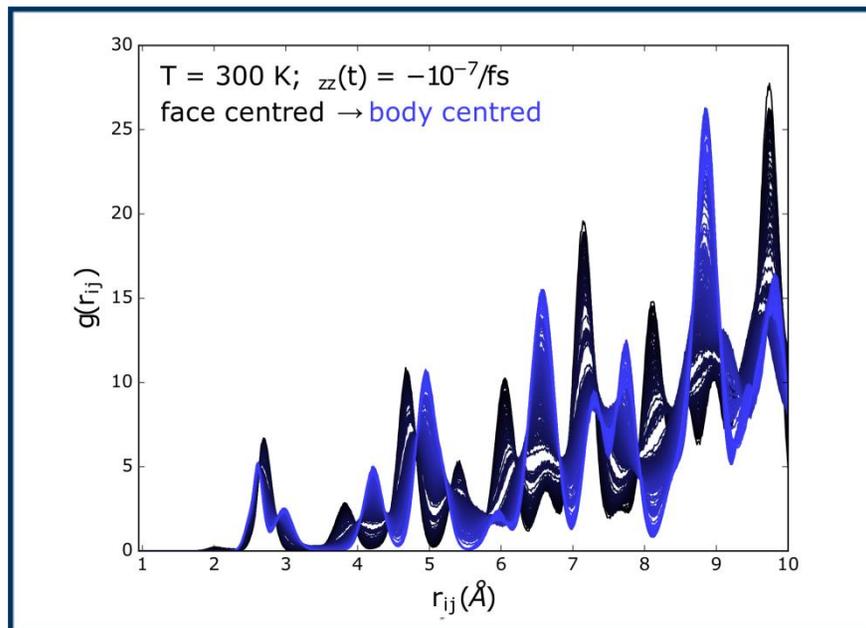
Setup according to the Bain path:

- Uniaxial deformation along the [001] direction with a constant strainrate of $\epsilon_{zz}(t) = -0.1 z_0/\text{ns}$
- 2 ns of simulated time
- Calculated RDF every 10 ps

RDF Structure Analysis

Radial Distribution Function

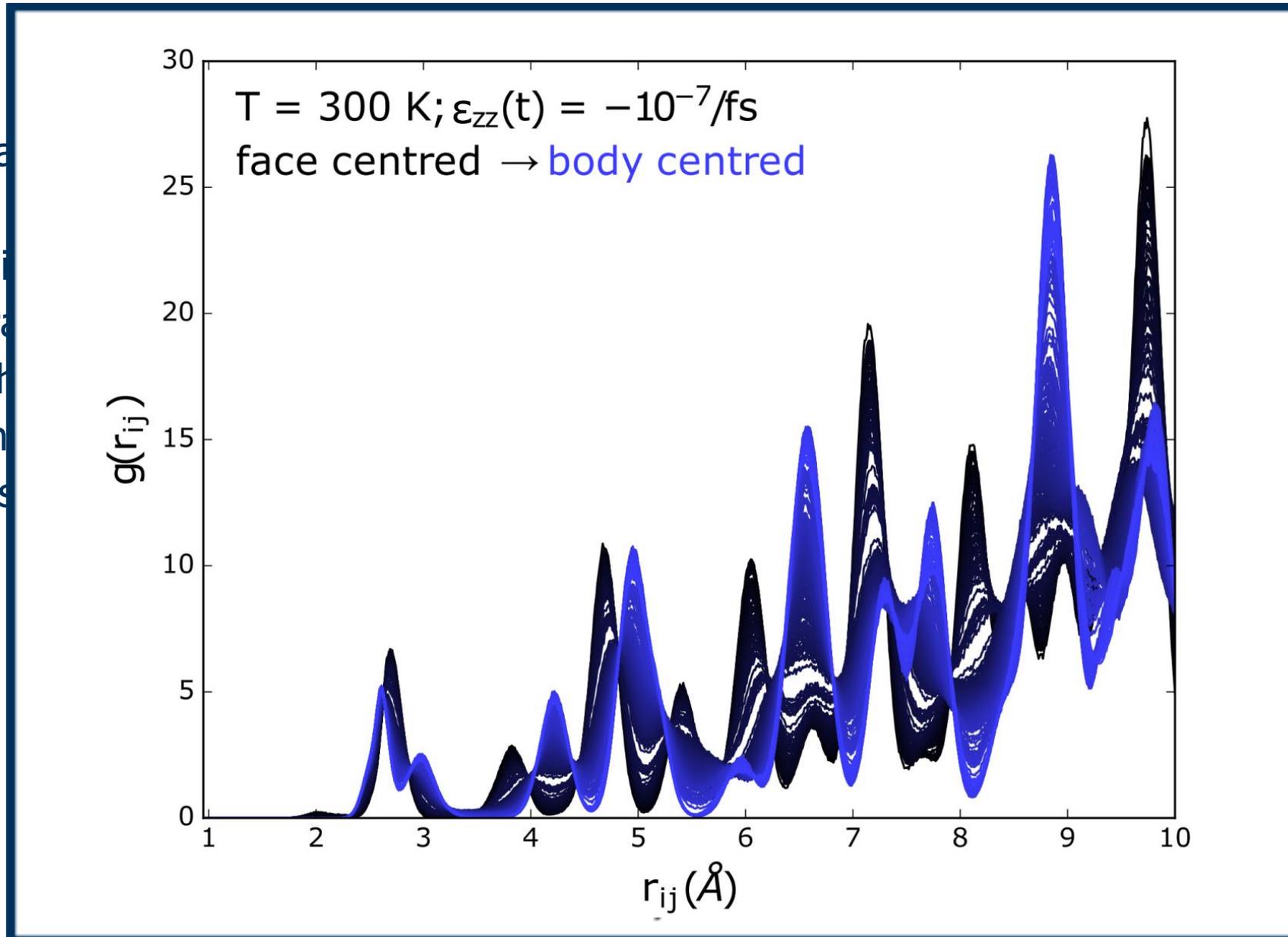
- Rather continuous transformation from *FC* zu *BC* structure
- Limited to resolve internal structures or events



Strain Induced Phase Transition

Radiation

- Limited tetrahedral
- Rather than transition BC sites



Strain Induced Phase Transition

Radial

- Limit
- tetra
- Rath
- tran
- BC s

$g(r_{ij})$

$g(r_{ij})$

T =
face

$g(r_{ij})$

$g(r_{ij})$

0

5

10

15

20

25

30

$r_{ij}(\text{\AA})$

1

2

3

4

5

6

7

8

9

10

$g(r_{ij})$

0

2

4

6

8

10

12

14

16

$r_{ij}(\text{\AA})$

1

2

3

4

5

6

7

8

9

10

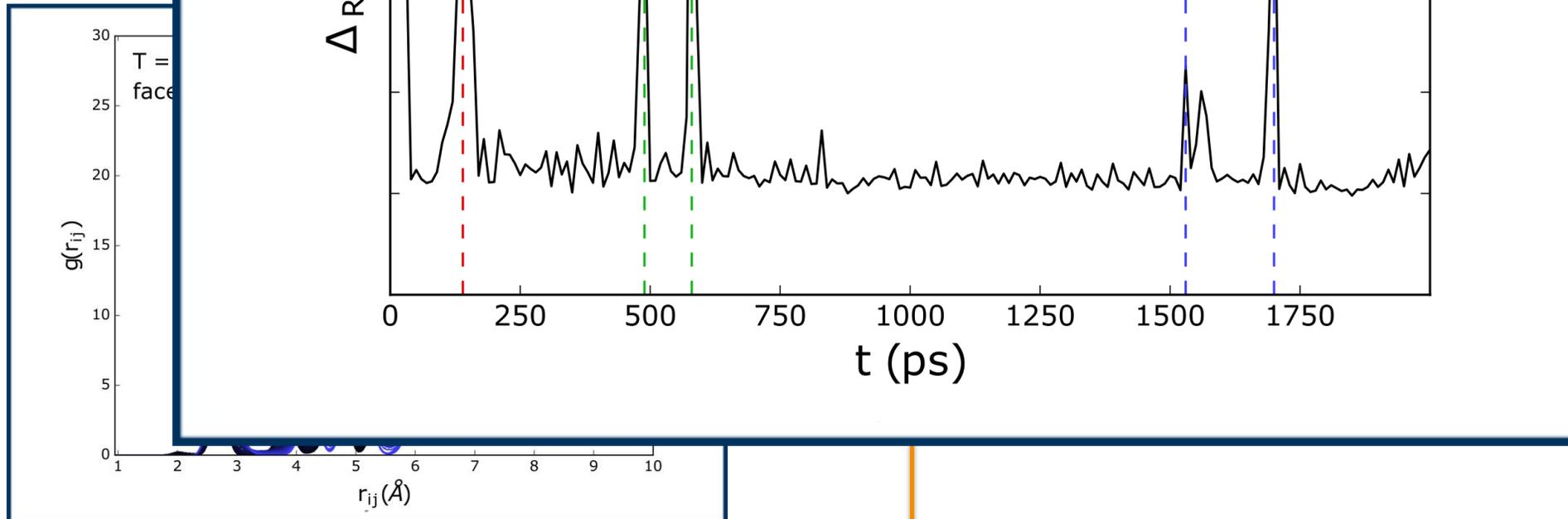
560ps.rdf
570ps.rdf

$$\Delta(r_{ij})_t = |g(r_{ij})_t - g(r_{ij})_{t-\alpha}|$$

Strain Induced Phase Transition

Radia

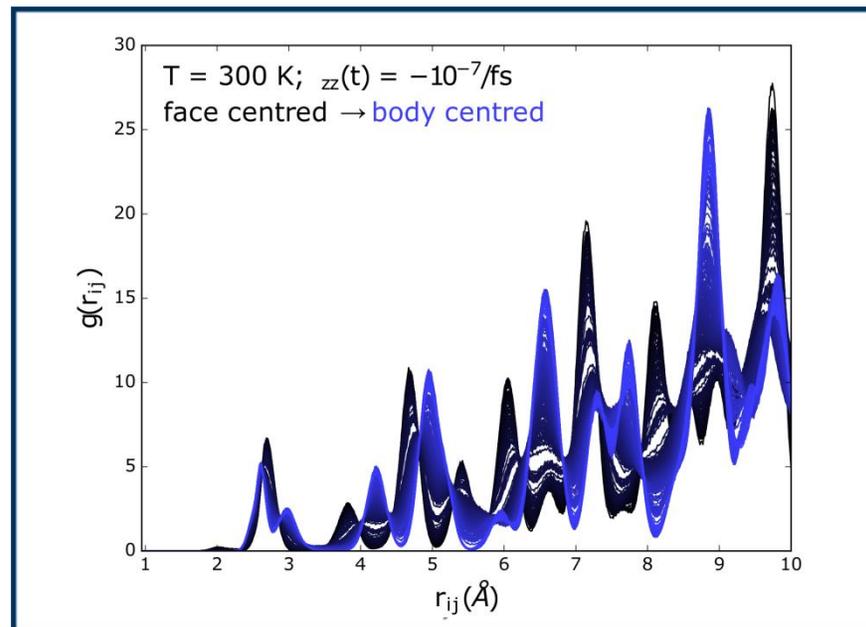
- Limit
- tetra
- Rath
- tran
- BC s



Structure Analysis

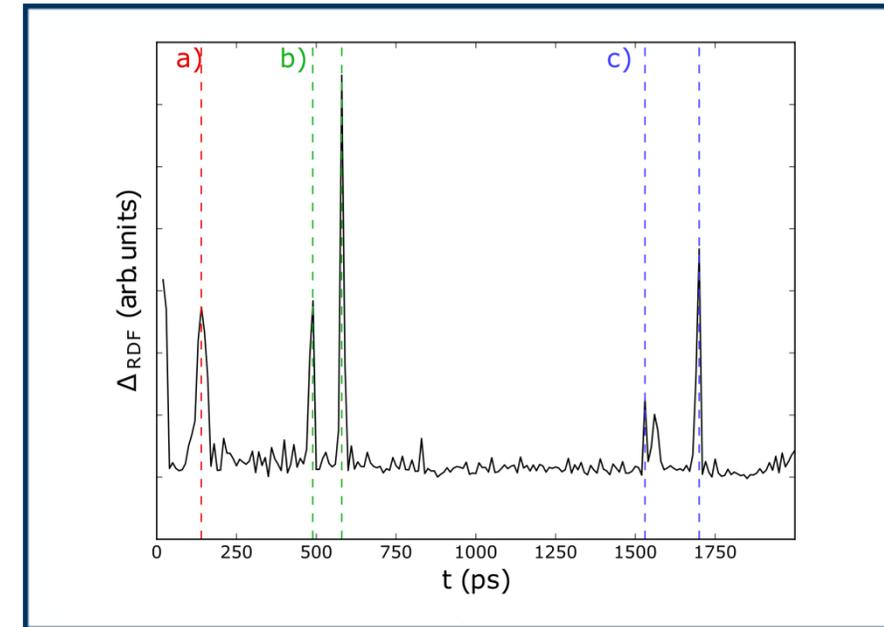
Radial Distribution Function

- Limited to resolve tetragonal phases
- Rather continuous transformation from *FC* zu *BC* structure



RDF Separation Function

- $\Delta_{RDF} = \int_{r_{ij}} |g(r_{ij})_{t_0} - g(r_{ij})_{t_0 -}| dr_{ij}$
- Indication of "avalanche" events



Common Neighbour Analysis

- Evaluation of lattice environment of atomic positions
- Output of the associated lattice type per atom

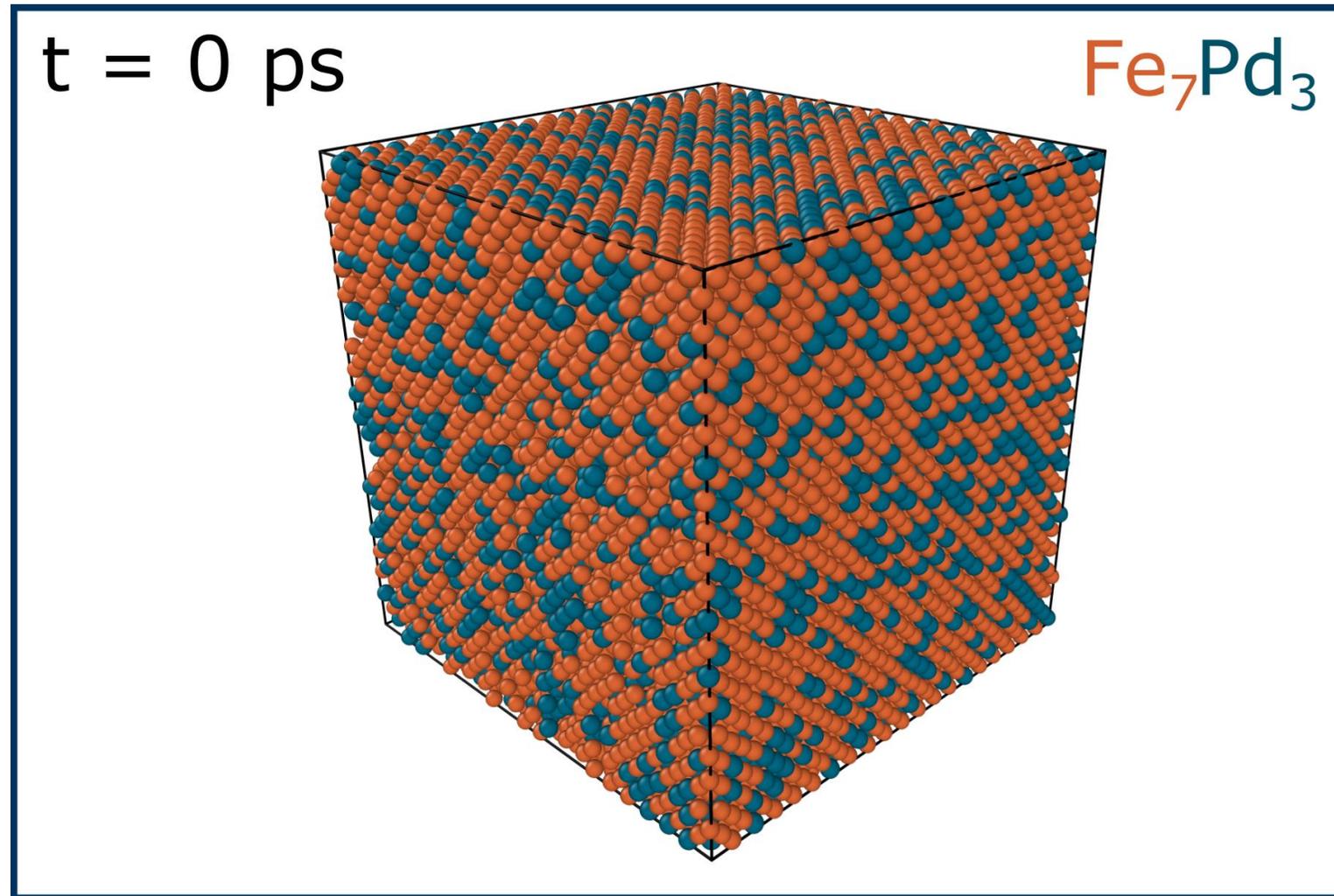
[Honeycutt and Andersen, J. Phys. Chem. 91, 4950]

Dislocation Analysis

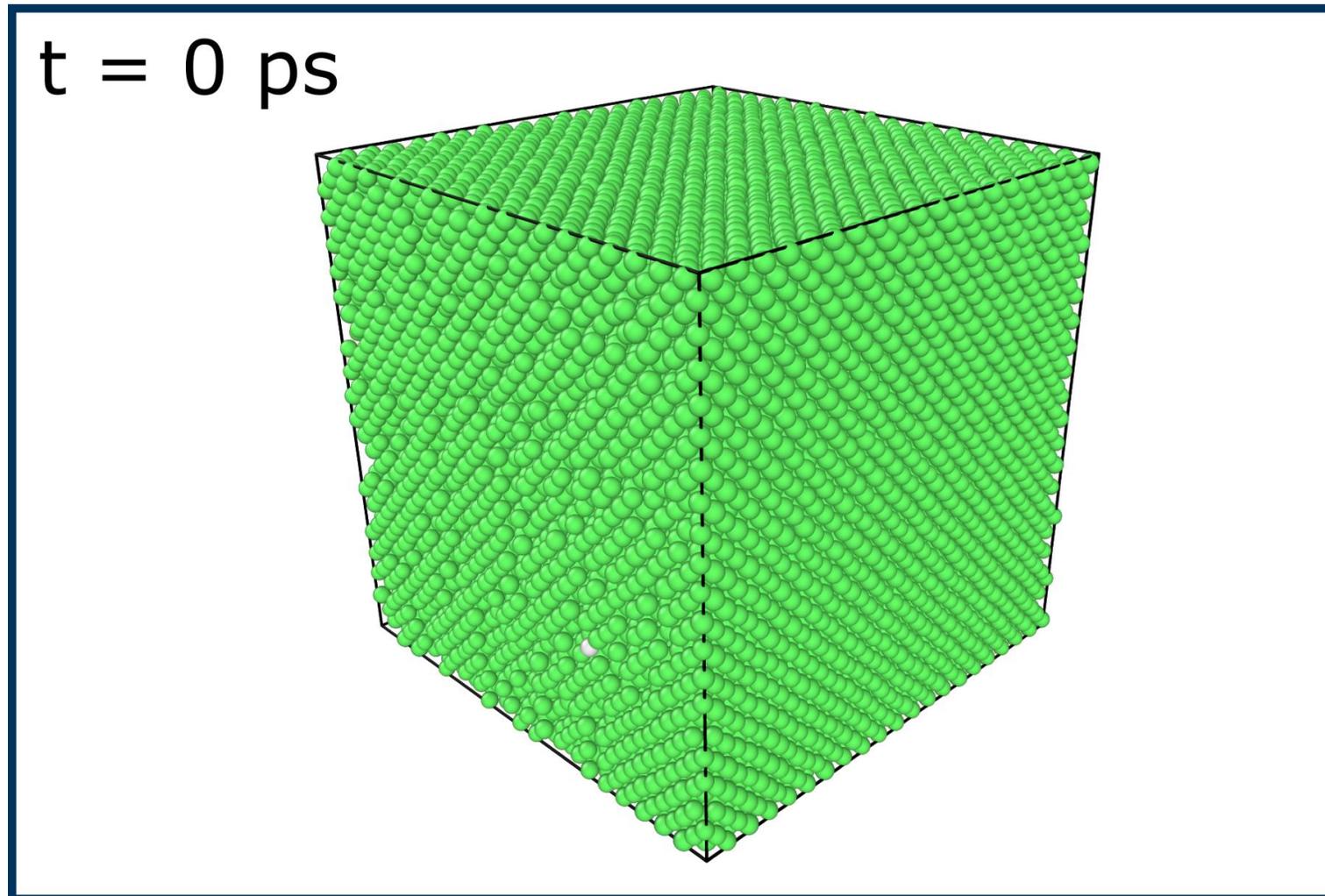
- Identification of all defects in a crystal of **given** lattice structure
- Output of defect mesh: triangulated surface deviding the "good" and the "bad" crystal region

[A. Stukowski, V.V. Bulatov and A. Arsenlis. Modelling Simul. Mater. Sci. Eng. 20, 085007 (2012)]

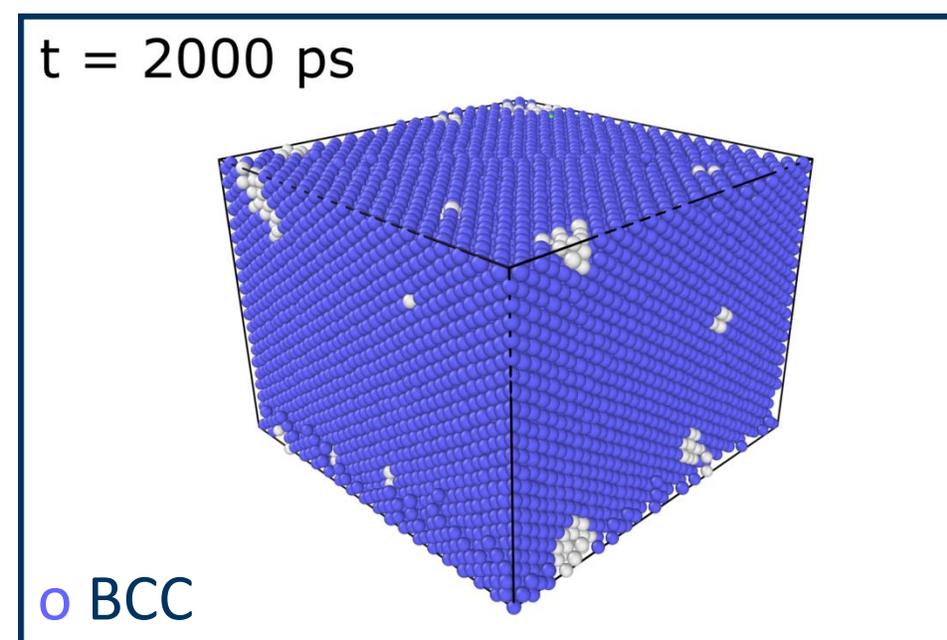
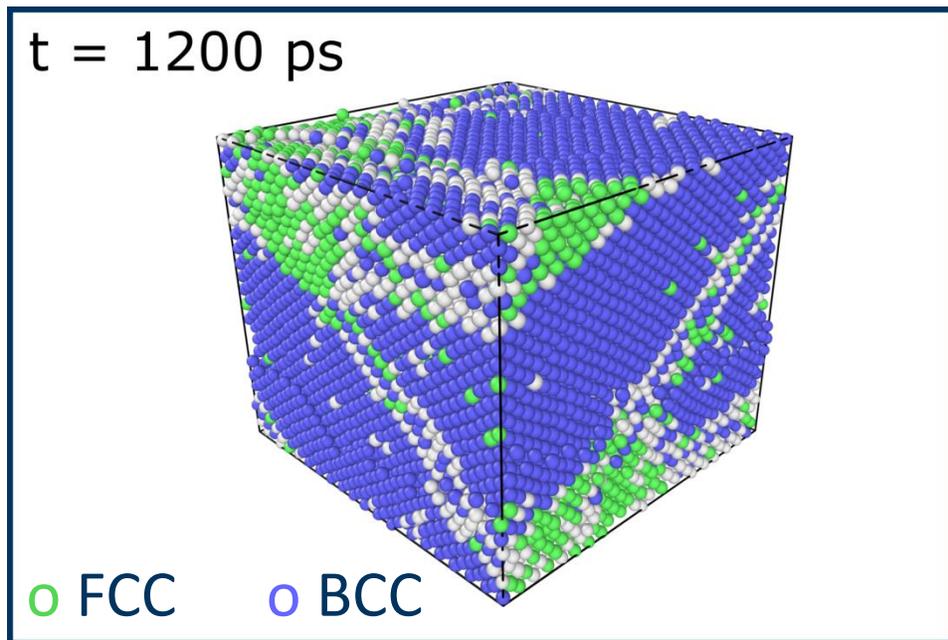
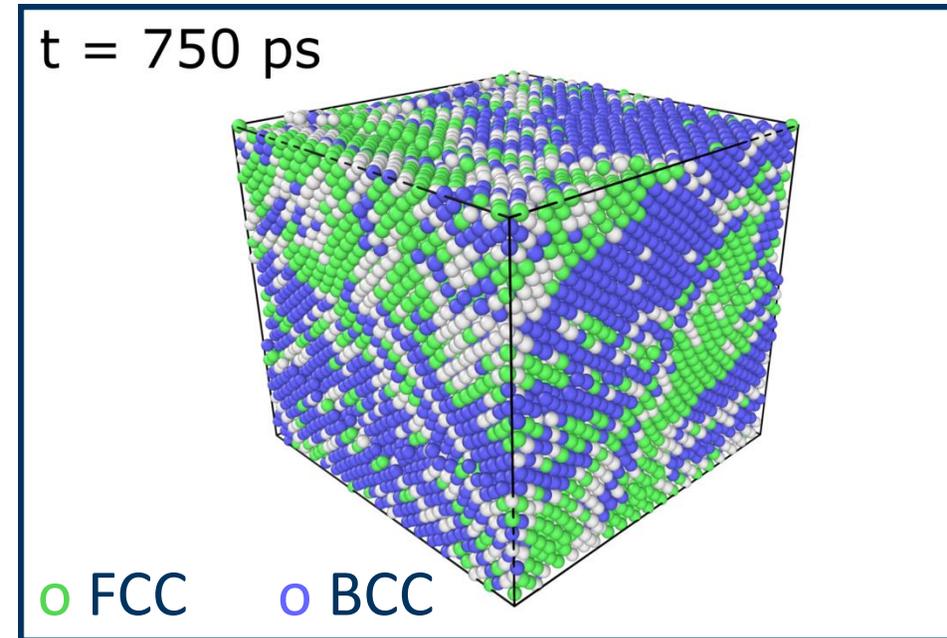
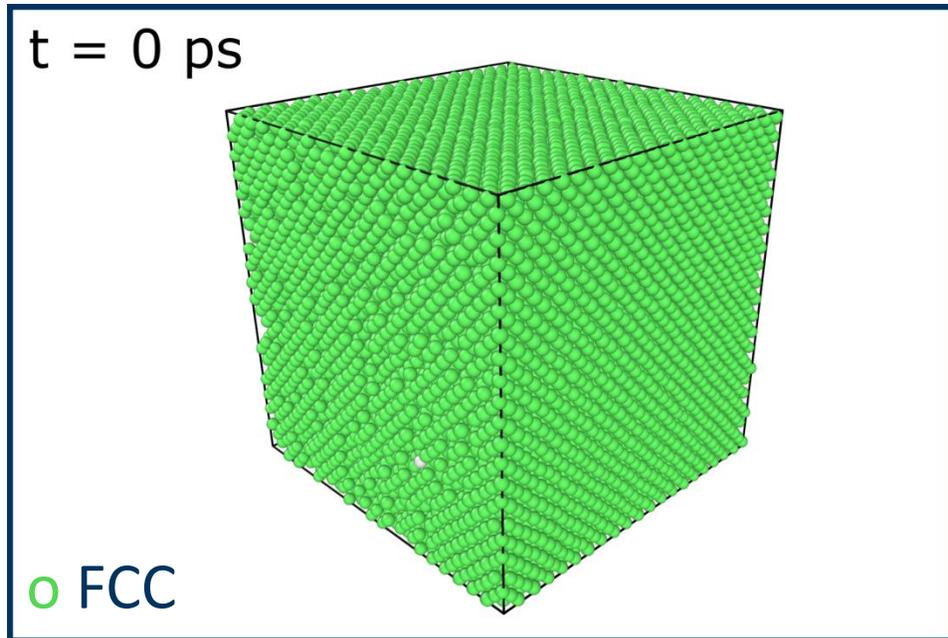
Strain Induced Phase Transition



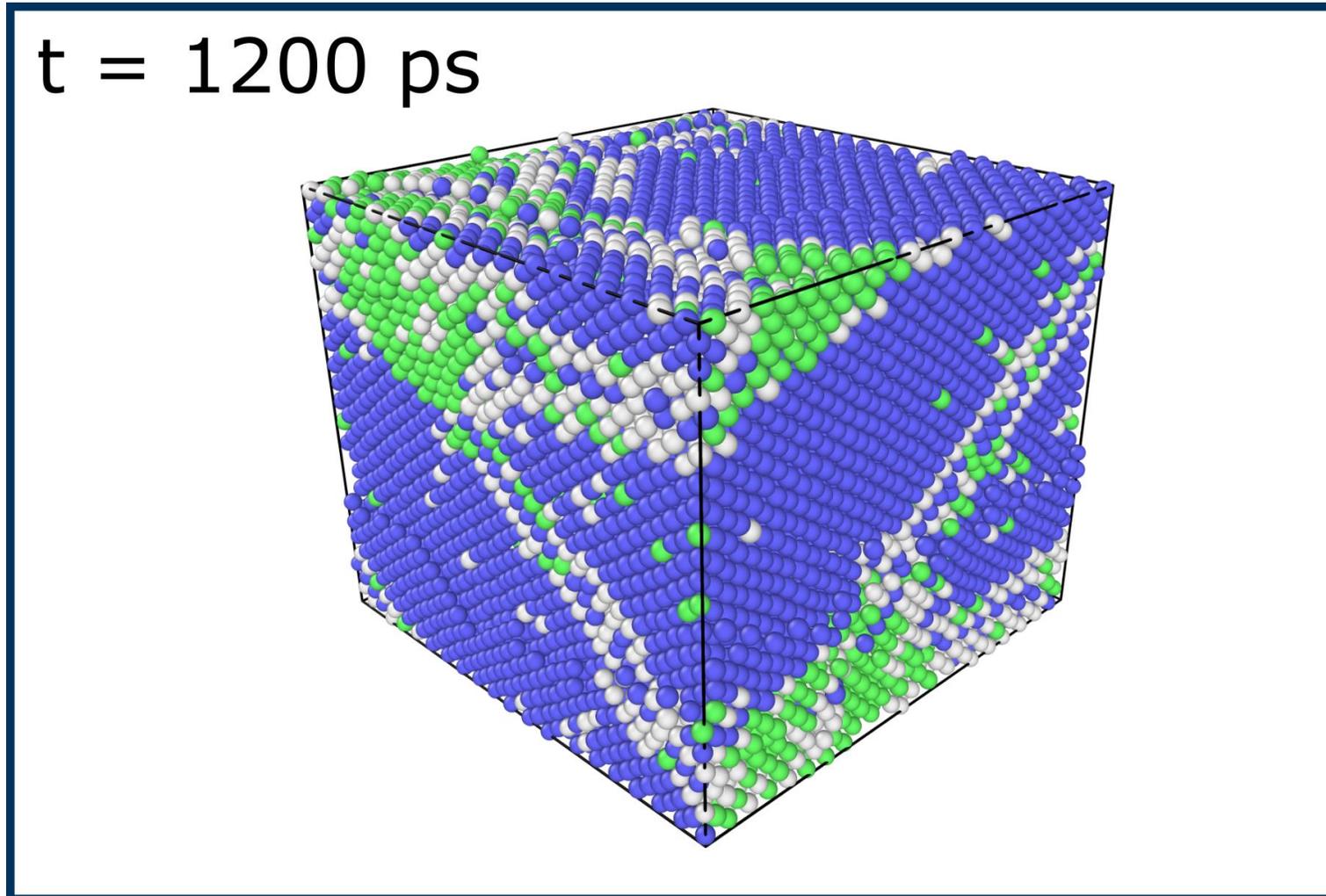
Strain Induced Phase Transition



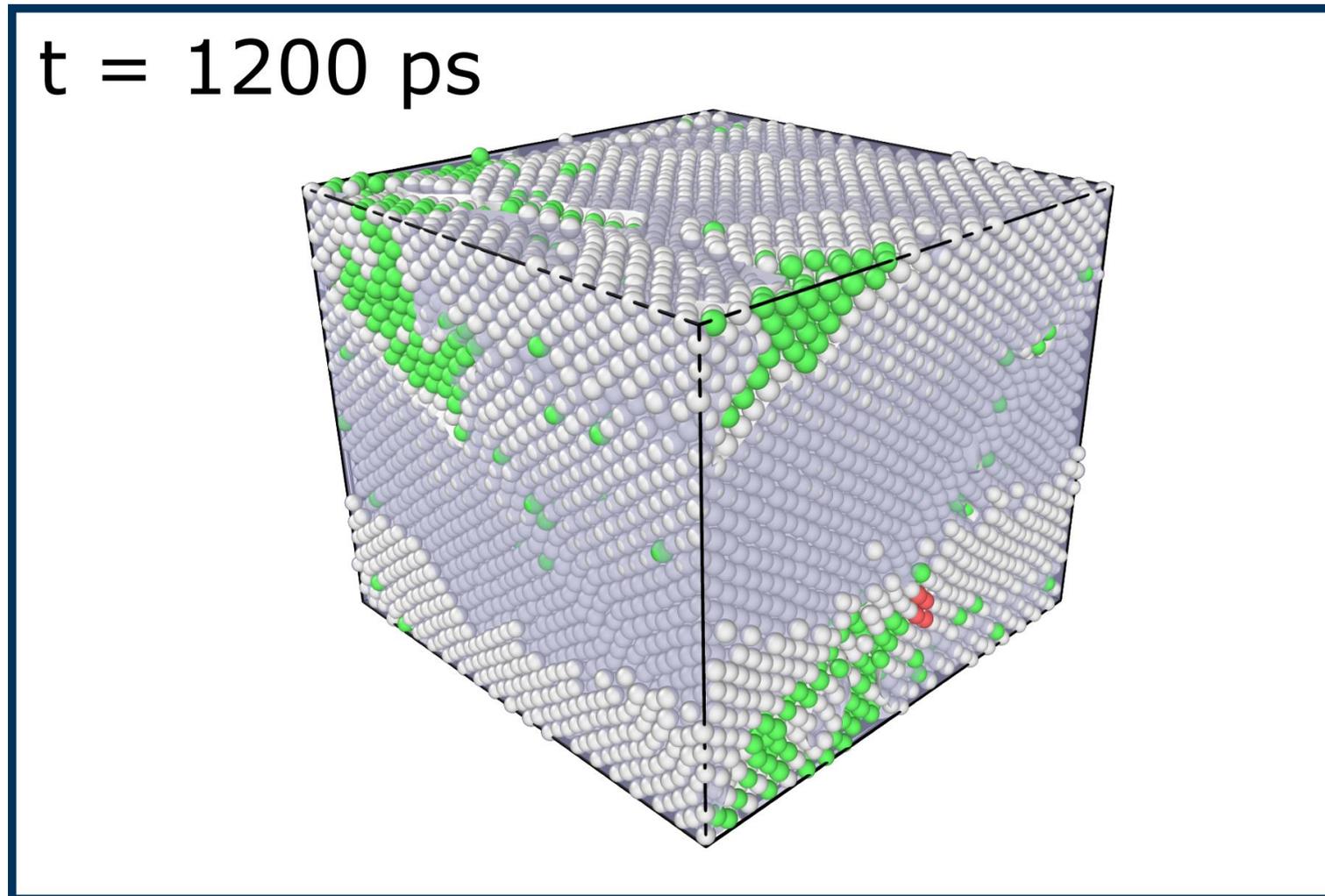
Strain Induced Phase Transition



Strain Induced Phase Transition

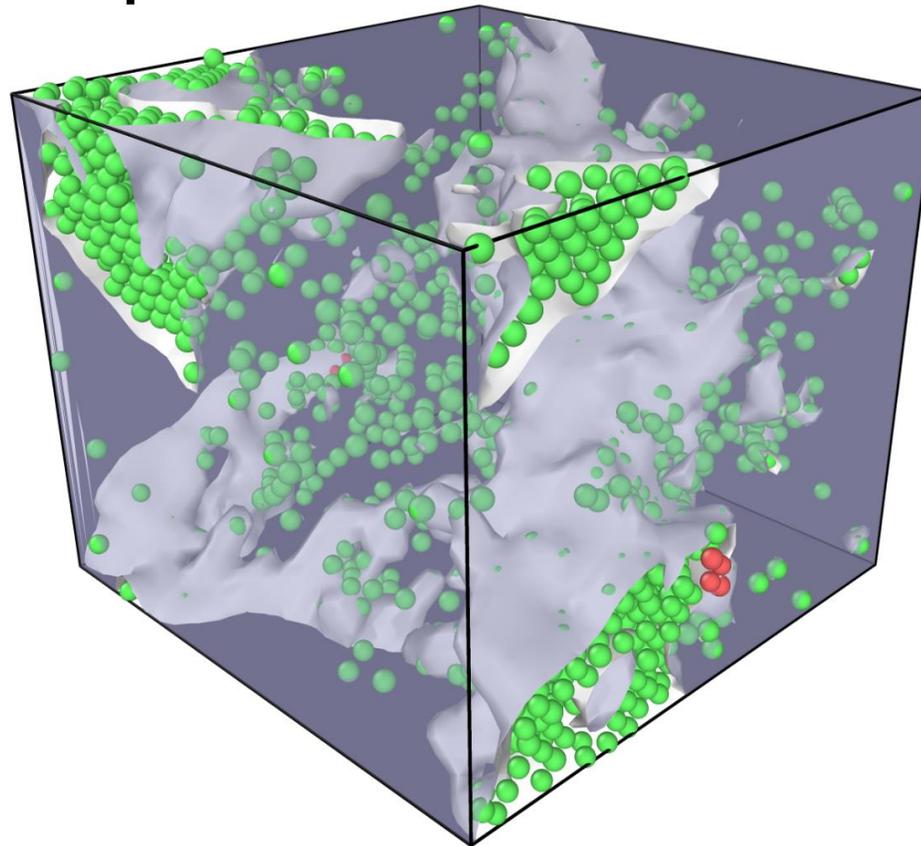


Strain Induced Phase Transition

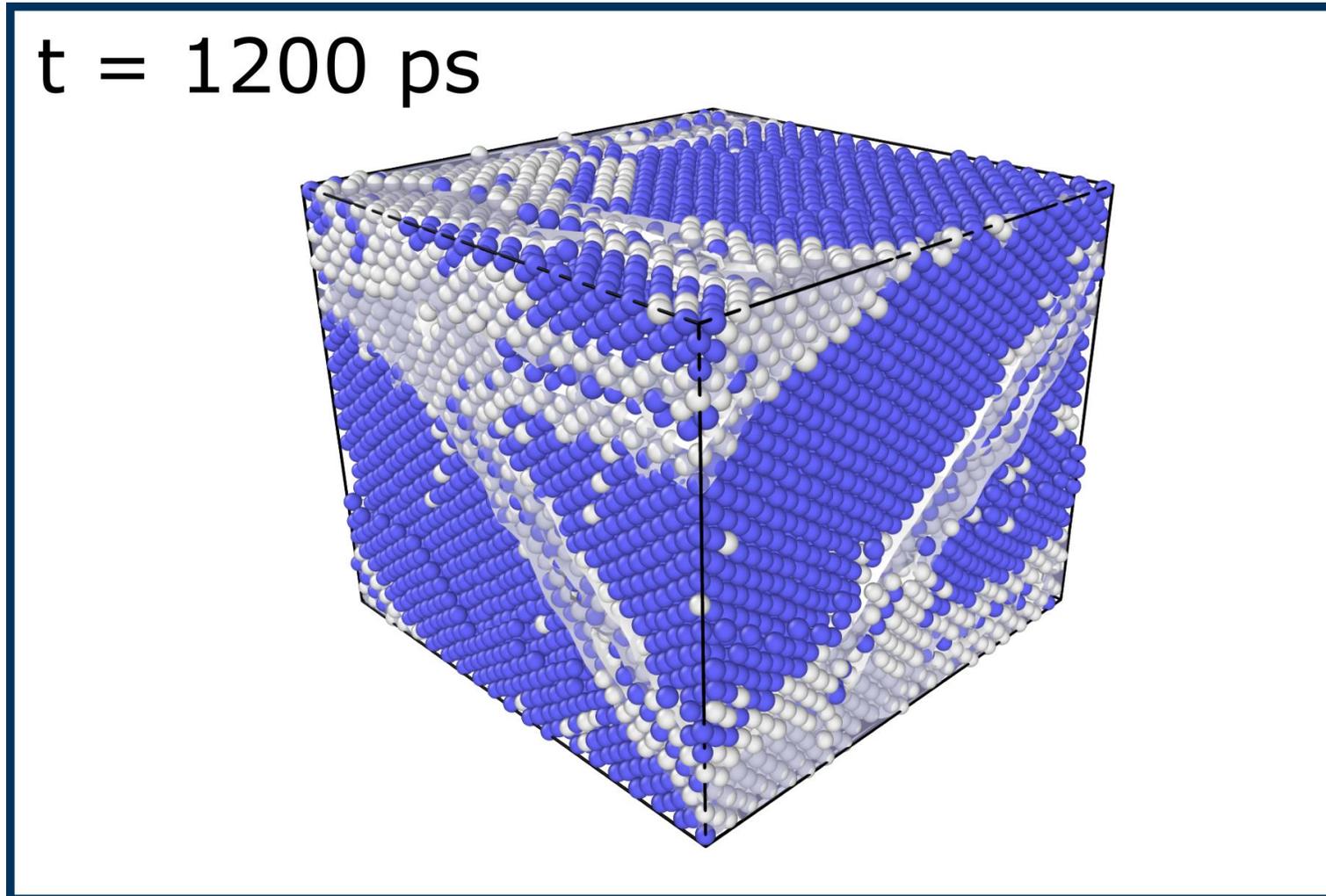


Strain Induced Phase Transition

$t = 1200 \text{ ps}$

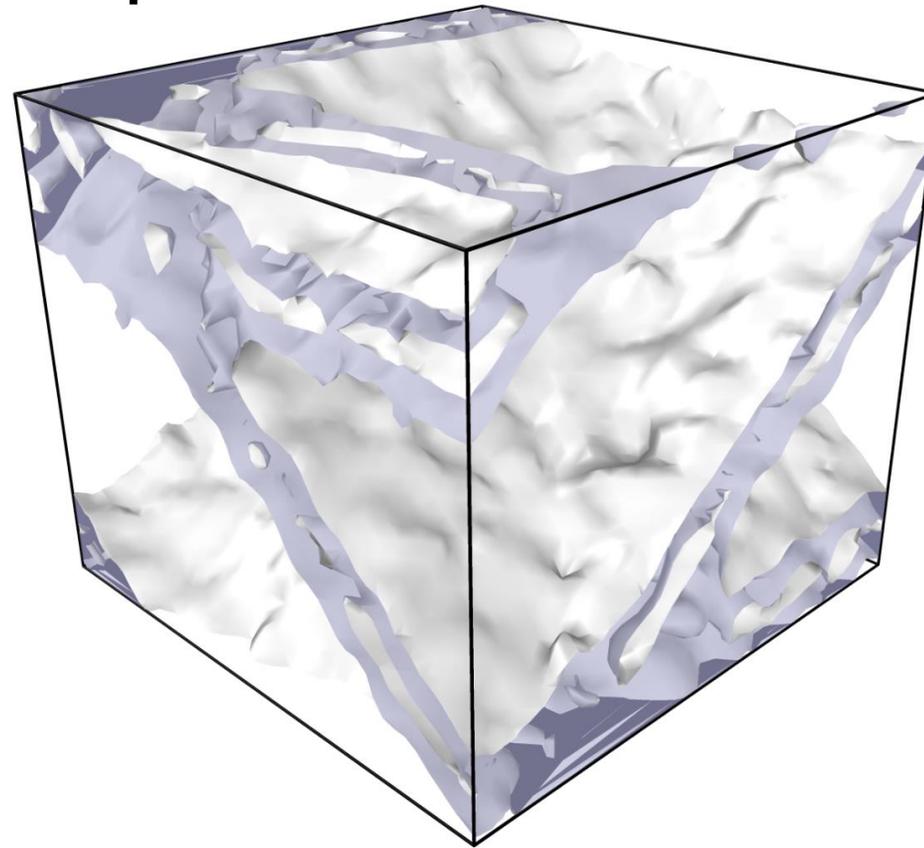


Strain Induced Phase Transition



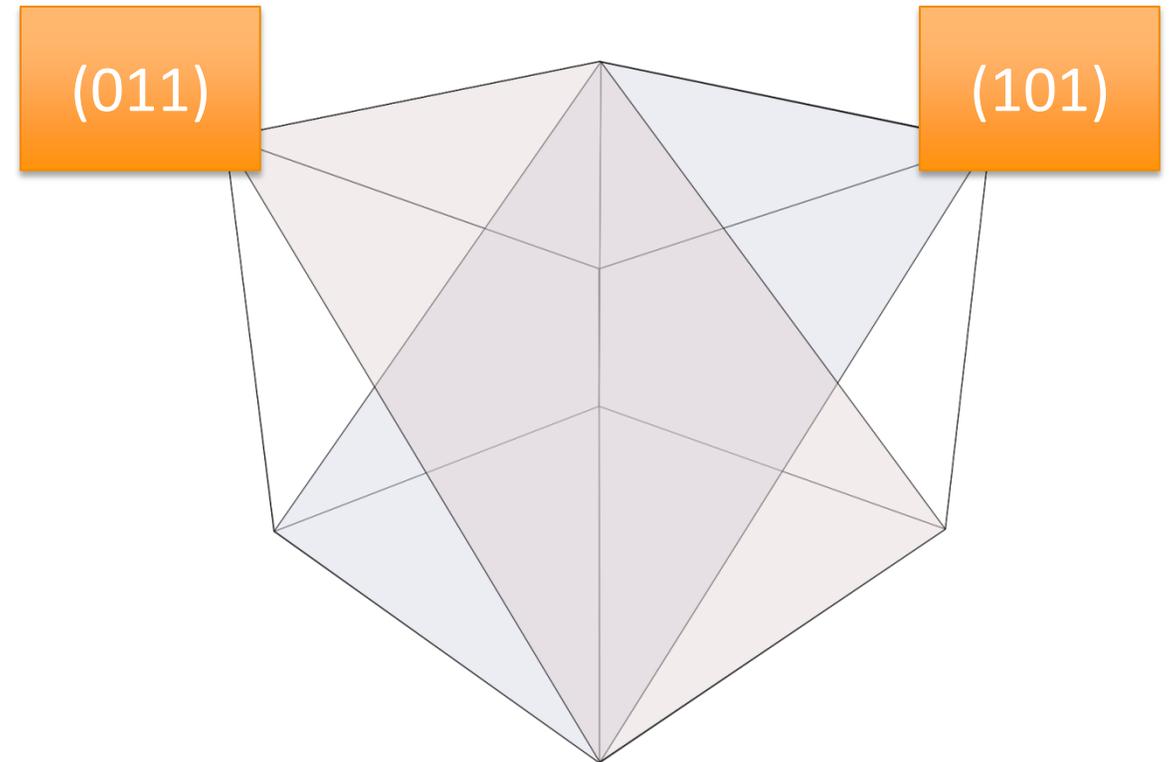
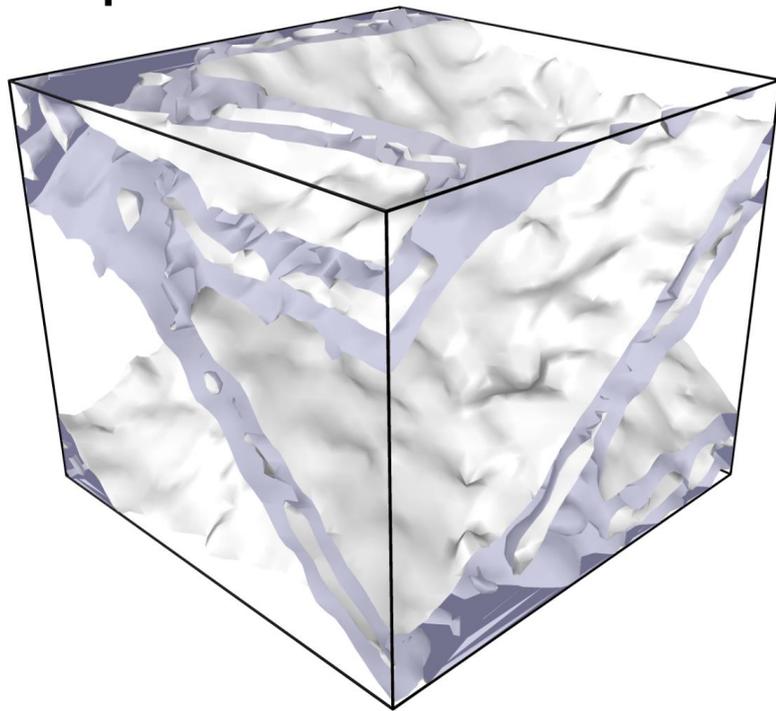
Strain Induced Phase Transition

$t = 1200 \text{ ps}$



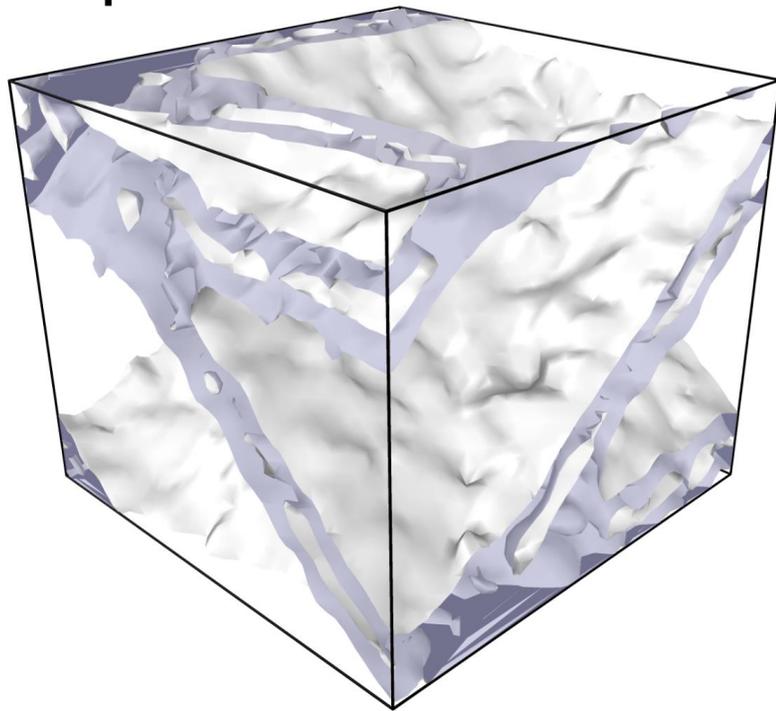
Strain Induced Phase Transition

$t = 1200 \text{ ps}$

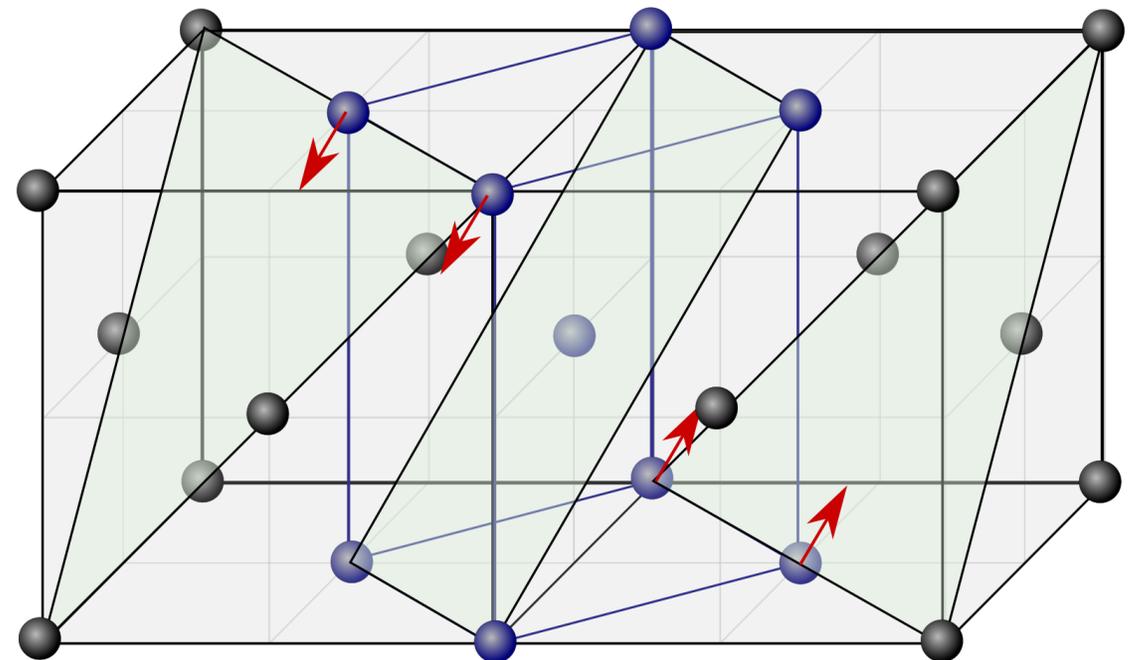


Nishiyama - Wassermann

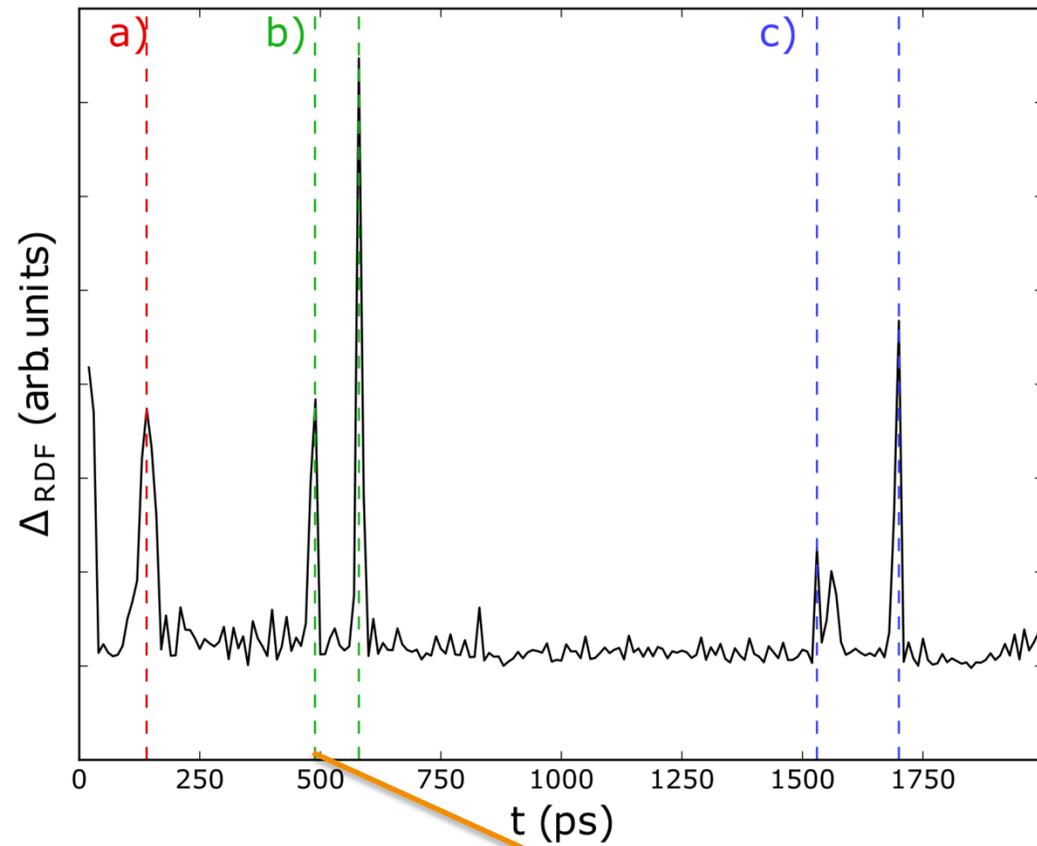
$t = 1200 \text{ ps}$



$(111)_{\text{FCC}} \parallel (011)_{\text{BCC}}$ $[11\bar{2}]_{\text{FCC}} \parallel [01\bar{1}]_{\text{BCC}}$

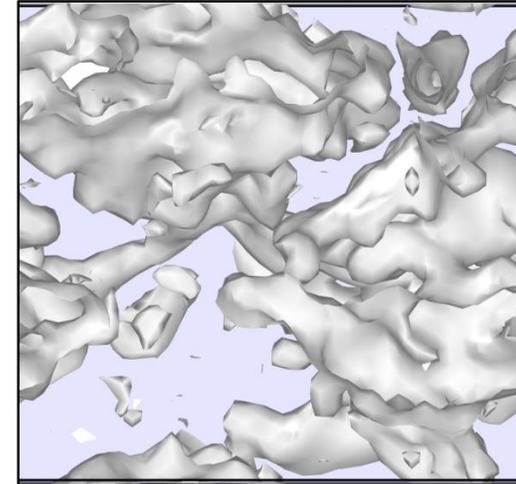


Analysing the Peaks

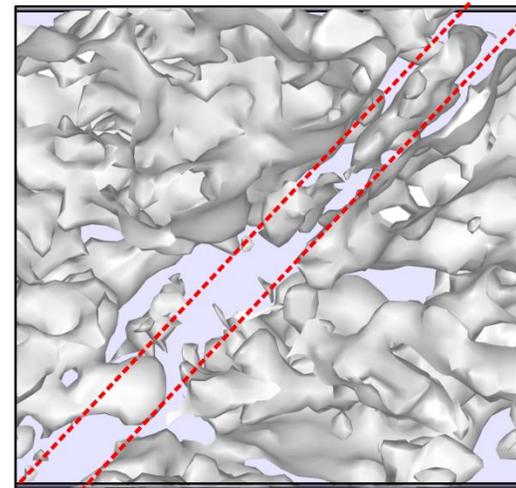


Formation of (101)
habit planes

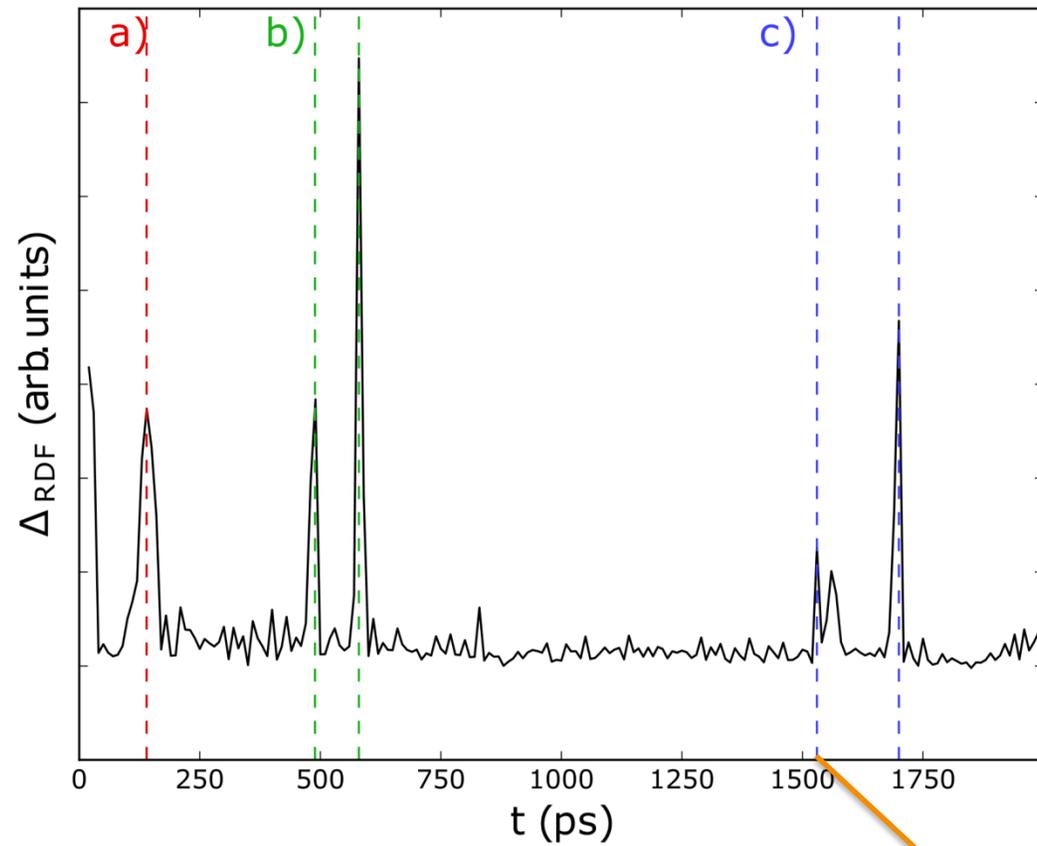
t = 470 ps View on y-plane



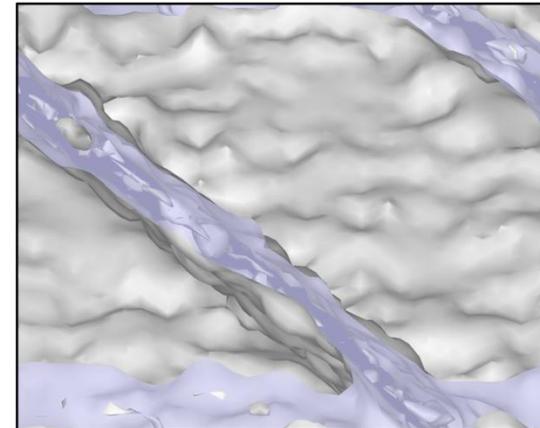
t = 480 ps



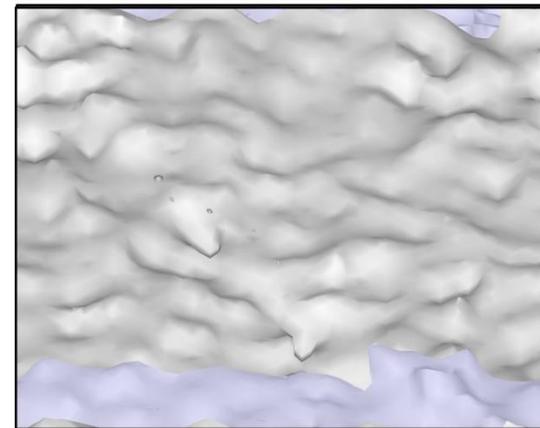
Analysing the Peaks



t = 1520 ps View on x-plane

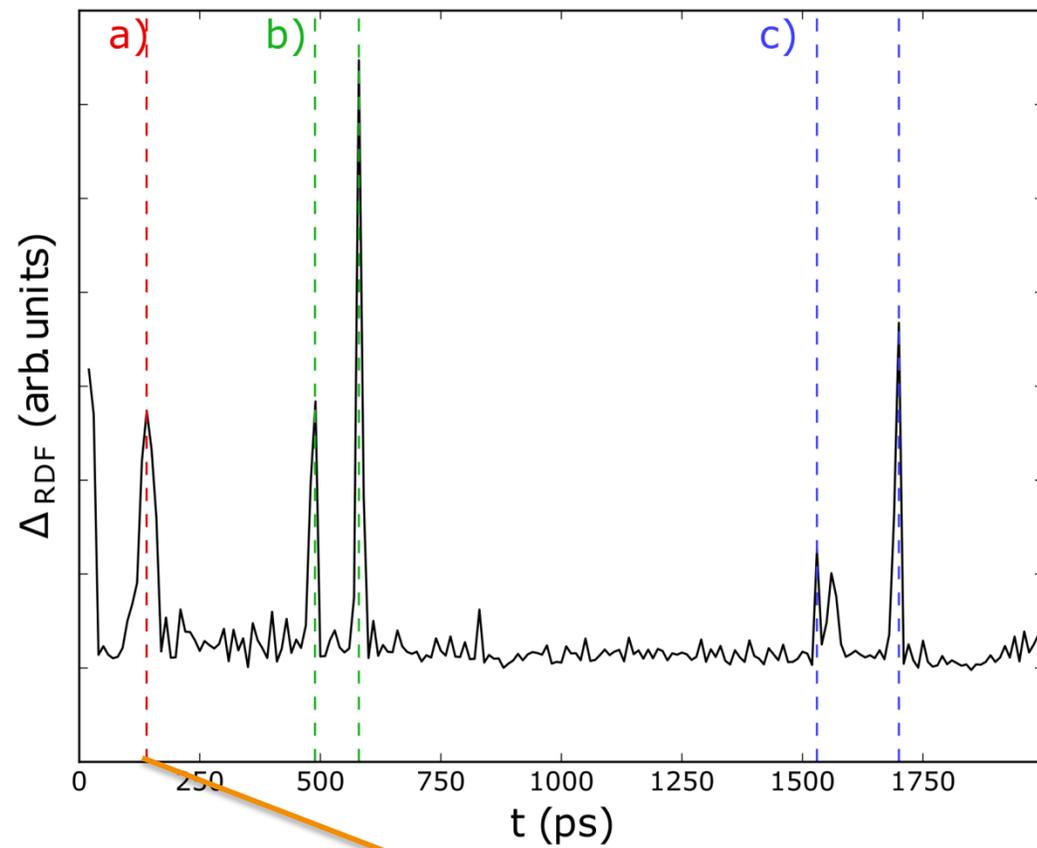


t = 1540 ps



Vanishing of (011)
habit planes

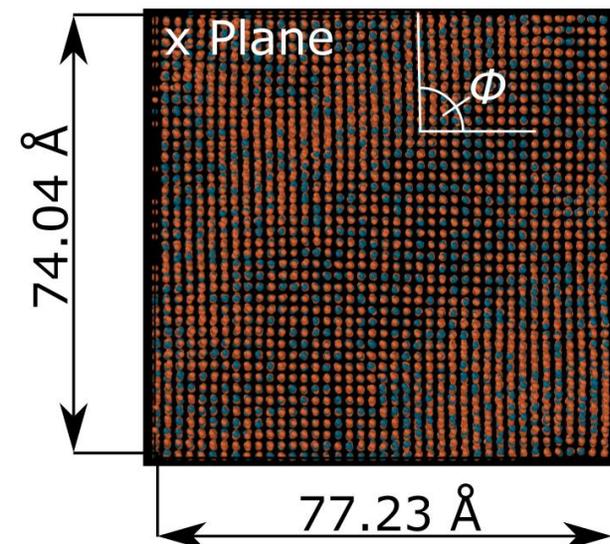
Analysing the Peaks



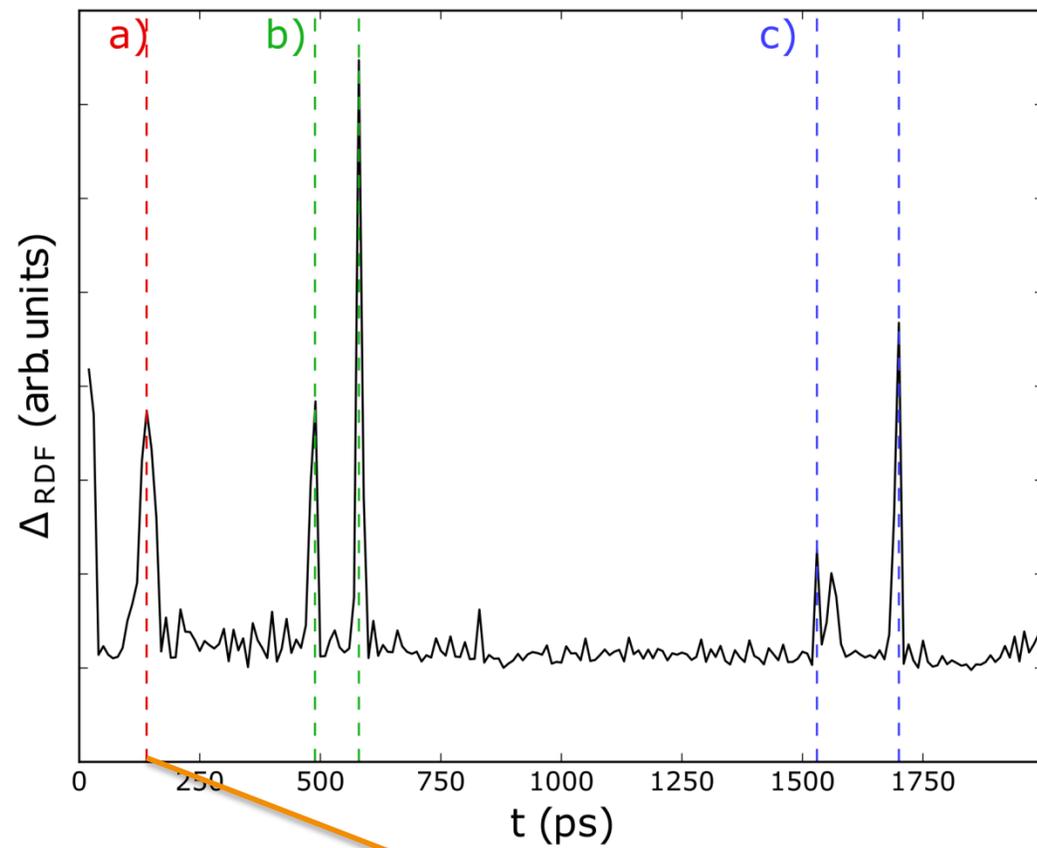
Angle of Tilt Dislocations

- $\varphi = 2 \left(\arctan \left(\frac{c_t}{a_t} \right) - \frac{\pi}{4} \right)$ [1]

[1] A. Khachaturyan, S. Shapiro, and S. Semenovskaya, Physical Review B 43, 10832 (1991).

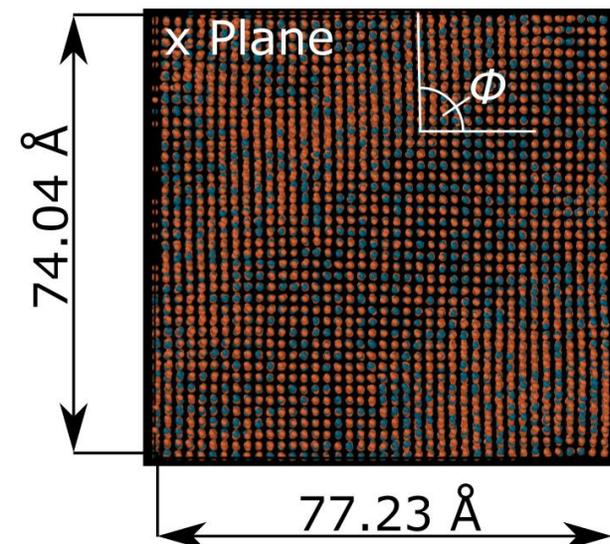


Analysing the Peaks

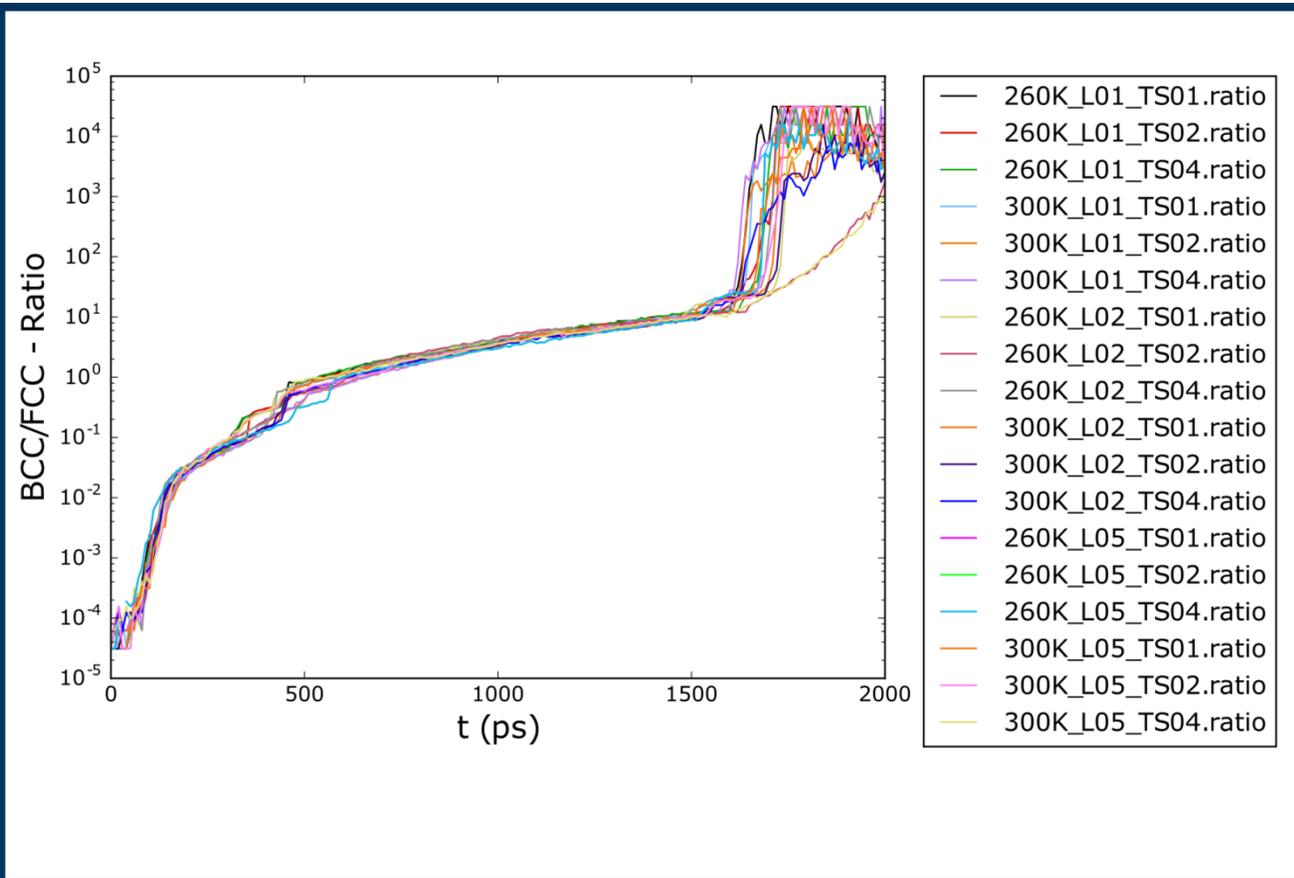


Angle of Tilt Dislocations

- $\varphi = 2 \left(\arctan \left(\frac{c_t}{a_t} \right) - \frac{\pi}{4} \right)$
- Indication of a temporary *FCT* phase



BCC/FCC Ratio



Check Bain Path for Different Temperatures, Temperature Sets and Atomic Orders

- Every simulation exhibits the same pattern of (101) and (011) habit planes
- *BCC/FCC* ratio graphs: all simulated transitions follow the same pattern (except for minor fluctuations)

Conclusion

- Successfully determined a phase transition temperature
- Nishiyama – Wassermann Transformation vs. Bain Path
- Avalanche effects:
 - Habit plane formation due to shear deformation
- RDF-Separation-Function complementary method to investigate structural changes

Acknowledgements

- Prof. Stefan G. Mayr
- “AG Mayr” work group
- Open Visualization Tool – “Ovito”
[www.ovito.org – Alexander Stukowski]
- Prof. Wolfhard Janke
- You!

Strain Induced Phase Transition

