Systematic Microcanonical Analyses of Polymer Adsorption Transitions

BuildMoNa



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The Model: Off-lattice Polymer near a Substrate Definitions for a Microcanonical Analysis

# Semiflexible 3*D*-Coarse-Grained Polymer near a Substrate

$$E_{\text{total}} = E_{\text{LJ}} + E_{\text{bend}} + E_{\text{sur}}$$



### Lennard-Jones Potential

$$E_{\rm LJ} = 4 \sum_{i=1}^{N-2} \sum_{j=i+1}^{N} \left( \frac{1}{r_{ij}^{12}} - \frac{1}{r_{ij}^{6}} \right)$$

#### Bending Energy

$$E_{\text{bend}} = \frac{1}{4} \sum_{k=1}^{N-2} \left( 1 - \cos\left(\vartheta_k\right) \right)$$

Surface Attraction  

$$E_{sur} = \epsilon_s \sum_{i=1}^{N} \left( \frac{2}{15} \frac{1}{z_i^9} - \frac{1}{z_i^3} \right)$$

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# Canonical vs. Microcanonical Approach

### canonical ensemble (controlled by intensive parameter T)

Def.: NVT-ensemble: statistical equilibrium ensemble, where the number of particles (N) and the volume (V) of each microstate are the same. The temperature (T) is the one of a common heat bath.

• resultant energy distribution: Boltzmann distribution  $p(E) = g(E)e^{-E/k_BT}$ 

#### microcanonical ensemble (controlled by extensive parameter E)

Def.: NVE-ensemble: ensemble of microstates with (N, V and E conserved), e.g., the system is isolated from any environment

- based on density of states g(E), that is still well-defined for non-extensive systems
- g(E) obtained by multicanonical Monte Carlo sampling

The Model: Off-lattice Polymer near a Substrat Definitions for a Microcanonical Analysis

# Important Definitions



#### microcanonical entropy

$$s(e) = N^{-1}k_B \ln g(e)$$
, with  $k_B = 1$  and  $e = E/N$ 

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Gibbs construction/adsorption temp.

concave hull  $\mathcal{H}_s(e) = s(e_{ads}) + e/T_{ads}$ , that touches s(e) at  $e_{ads}$  and  $e_{des}$ 

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Latent heat per monomer required to break the surface contacts at  $T_{\rm ads}$ 

$$\Delta q = e_{\rm des} - e_{\rm ads} = T_{\rm ads}[s(e_{\rm des}) - s(e_{\rm ads})]$$

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#### Surface entropy

 $\Delta s_{\text{surf}} \equiv \Delta s(e_{\text{sep}})$ , where  $e_{\text{sep}}$  is the energy with  $\Delta s(e) = \mathcal{H}_s(e) - s(e)$  being maximal

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The Origin of the Peak Increasing Substrate Attraction Increasing Chain Length Scaling of the Convex Intruder Adsorption Temperature

Let's start simple: Density of States g(E) of a Free 3mer

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N=3 and  $\epsilon_s=0$ 

Here the hamiltonian can be written as:

$$H(\vartheta) = E_{\rm LJ} + E_{\rm bend}$$
  
= 4(2 + 2 cos \vartheta)^{-3}((2 + 2 cos \vartheta)^{-3} - 1)  
+ (1 - cos \vartheta)/4



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Density of States g(e) of a Free Polymer increasing

the Chain Length



The Origin of the Peak Increasing Substrate Attraction Increasing Chain Length Scaling of the Convex Intruder Adsorption Temperature

Microcanonical Entropy  $s(e) = \ln g(e)$  of a Free Polymer

increasing the Chain Length



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## Now add and increase a Substrate Attraction

- for low  $\epsilon_s$  no convex intruder can be found (continuous adsorption, docking)
- $\bullet\,$  with increasing  $\epsilon_s$  the amount of low-energy states increases and s(e) gets convex at some  $\epsilon_s$
- $T_{\rm ads}$  (inverse slope),  $\Delta q$  and  $\Delta s_{\rm surf}$  increases with  $\epsilon_s$



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## Relation between scaling and conformation



- for  $\epsilon_s \gtrsim 2$  the adsorption transition takes place at higher T than the  $\Theta$ -transition and scaling converges
  - $\rightarrow$  scaling directly linked to conformations?



M. Möddel, M. Bachmann, and W. Janke, J. Phys. Chem. B **113**, 3314 (2009). Basic Definitions Simulation Results Conclusions Concl

## And what happens for longer Polymers?

- the maximum in s(e) disappears for long chains
- $\Delta s_{\text{surf}}$  as well as  $\Delta q$  decrease with N, while  $T_{\text{ads}}$  seems to saturate
- slope for high e increases with N (microcanonical temperature changes sign)





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The Origin of the Peak Increasing Substrate Attraction Increasing Chain Length Scaling of the Convex Intruder Adsorption Temperature

# Scaling of the Convex Intruder for different $\epsilon_s$

 $\Delta s_{\text{surf}}$  and  $\Delta q$ 

the data clearly suggest

- $\Delta s_{\text{surf}} \propto N^{-\kappa_s}$
- $\Delta q \propto N^{-\kappa_q}$

for  $\epsilon_s \gtrsim 2$ 

$\epsilon_s$	$\kappa_s$	$\kappa_q$
3	$1.647\pm0.014$	$0.390 \pm 0.004$
4	$1.360\pm0.013$	$0.368 \pm 0.004$
5	$1.237\pm0.008$	$0.367 \pm 0.003$
6	$1.166\pm0.005$	$0.358 \pm 0.004$

for ε<sub>s</sub> ≥ 4 single layer conformations get likely at low T and Δq ∝ ε<sub>s</sub>



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#### for comparison: First order: $\Delta s_{\text{surf}} \sim N^{-1}$

 $\Delta q - \Delta q_{\epsilon_s \to \infty} \sim N^{-1}, \ \Delta q_{\epsilon_s \to \infty} \neq 0$ here:  $\Delta q_{\epsilon_s \to \infty} = 0$ 

different scaling  $\rightarrow$  continuous transition



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Saturation of the Adsorption Temperature with Chain Length

- $T_{\rm ads}$  monotonically increasing with N
- $\bullet\,$  likely not diverging, but reaching a fixed (probably universal) value for  $N\to\infty$



# Conclusions

- Adsorption transition exhibits clear signals of a first-order-like conformational transition in the important case of finitely long polymers
  - expressed by coexistence of adsorbed and desorbed conformations at the adsorption temperature
- A better understanding can be obtained by separating s(e) in adsorbed and desorbed part or deriving it for very short chains
- The transition crosses over into a second-order phase transition in the thermodynamic limit, as expected
- Scaling of  $\Delta s_{\text{surf}}$  and  $\Delta q$  was found for not too small  $\epsilon_s$
- Approach also feasible for a study of simulation box/density-dependence

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# **Bonus** Material

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