Autocorrelation Study for a Coarse-Grained Polymer Model

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Autocorrelation Function and Autocorrelation Time

Autocorrelation Function:

- $A(k) = \frac{\langle O_m O_{m+k} \rangle \langle O_m \rangle^2}{\langle O_m^2 \rangle \langle O_m \rangle^2}$ where *O* is an observable, *m* indicates any point in the time series
- Describe the correlation between two measurements with time displacement *k*

Exponential Autocorrelation Time:

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$$A(k) \stackrel{k \to \infty}{\longrightarrow} e^{-k/\tau_{O, exp}}$$

- $\tau_{O, \exp}$ is the exponential autocorrelation time
- For large k, behavior is non-exponential. (statistics issue)



Integrated Autocorrelation Time:

Since A(k) ^{k→∞}→ 0, the estimator will finally converge to a constant.

Binning Analysis:

- $\tilde{\tau}_{O,\text{bin}}(N_B) = N_B \tilde{\sigma}_{\overline{O}^B,c}^2 / 2\tilde{\sigma}_O^2$ where N_B is the binning block size, $\tilde{\sigma}_{\overline{O}^B,c}^2$ is the variance of an individual binning block average and $\tilde{\sigma}_O^2$ is the variance of an individual measurement
- As N_B increases, $\tilde{\tau}_{O,\text{bin}}$ will converge to $\tilde{\tau}_{O,\text{int}}$.



Model

- Number of Monomers: $N_m = 30,55$
- Energy for Conformation ζ :

$$E(\zeta) = \sum_{i=1}^{N-2} \sum_{j=i+2}^{N} V_{LJ}^{mod}(r_{ij}) + \sum_{i=1}^{N-1} V_{FENE}(r_{ii+1})$$

Non-bonded Lennard-Jones Potential:

$$V_{\text{LJ}}^{\text{mod}}(r_{ij}) = V_{\text{LJ}}(r_{ij}) - V_{\text{LJ}}(r_c)$$

with

$$V_{\mathsf{LJ}}(r_{ij}) = 4 \left[\left(\frac{\sigma}{r_{ij}} \right)^{12} - \left(\frac{\sigma}{r_{ij}} \right)^{6} \right]$$

where $\sigma = r_0/2^{1/6}$ ($r_0 = 1$) and cut off distance $r_c = 2.5\sigma$

FENE Potential:

$$V_{\text{FENE}}(r_{ii+1}) = -\frac{\kappa}{2}R^2 \ln\left[1 - \left(\frac{r_{ii+1}-r_0}{R}\right)^2\right]$$

where $R = 3/7$ and $K = 19.6$

Quantities:

 $\begin{array}{l} \mbox{Energy: } \langle E \rangle \\ \mbox{Number of Contacts: } \langle N_c \rangle \\ \mbox{Square of End to End Distance:} \\ \langle R^2_{ee} \rangle = \langle (\vec{r}_{N_m} - \vec{r}_1)^2 \rangle \\ \mbox{Square of Radius of Gyration:} \\ \langle R^2_{gyr} \rangle = \langle \frac{1}{N_m} \sum_{i=1}^{N_m} (\vec{r}_i - \vec{r}_{c.o.m.})^2 \rangle \end{array}$

 $(\vec{r}_{c.o.m.}$ is the position of the center of mass)



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Algorithm

• Algorithm:

Metropolis Monte Carlo method

• Update Method:

random local displacements of single monomers (i.e., random move in a small cubic box with edge length $d = 0.3r_0$)

• Reasons:

Metropolis algorithm works in canonical ensemble and generates the true physical distribution (Boltzmann distribution).

Displacement update is similar to the Brownian motion process.

"Phase" Transitions for a 30-mer



- 2nd order "phase" transition)
- $T \approx 1.4$, random-coil phase \longleftrightarrow globular phase (collapse transition, 2nd order "phase" transition)

Autocorrelation Times for a 30-mer and a 55-mer



- calculate autocorrelation times at 17 temperatures
- Curve fitting: $f_O(x) = \tau_O^f (1 - e^{-x/x^f})$ where x is N_B/k_{max} ; τ_O^f and x^f are two fit parameters



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30-mer case:



- Autocorrelation time curves estimated from two methods are identical.
- Autocorrelation times diverge near T = 0.3 (freezing transition); slowing down occurs at $T \approx 1.4$ (collapse transition).

55-mer case:



• The same curves appear in the 55-mer case except that all the values become larger.

- We use Metropolis algorithm combined with displacement update to do the simulation for a coarse-grained flexible polymer model.
- By using two methods, we've constructed the $\tilde{\tau}_{int}$ vs. T and $\tilde{\tau}_{bin}$ vs. T graphs.
- Due to slowing down, an extremal autocorrelation time can be considered as an indicator for the collapse transition.
- This is particularly useful for finite systems, where response quantities do not necessarily exhibit clear indications for pronounced thermal activity.

K. Qi and M. Bachmann, Autocorrelation Study for a Coarse-Grained Polymer Model, preprint.

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Thank you! Questions?





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